Ethanol oxidation activity and structure of carbon-supported Pt-modified PdSn-SnO₂ influenced by different stabilizers

Hui Wang, Ziyue Liu, Shan Ji, Keliang Wang, Tianbao Zhou and Rongfang Wang

Abstract

PdSn-SnO₂ nanoparticles supported on Vulcan XC-72 carbon were synthesized by chemical reduction in the presence of three different stabilizing agents: ethylene diamine tetra-acetic acid (EDTA), sodium citrate (Nacitrate) and hexamethylenetetramine (HMTA). TEM analysis showed that PdSn-SnO₂ /C catalyst made using the HMTA stabilizer produced the smallest particle size. XRD analysis detected the presence of PdSn alloy and the SnO₂ phase in all three PdSn-SnO₂ /C samples, and showed that PdSn-SnO₂ (HMTA) had the smallest lattice parameter. After PdSn-SnO₂ samples were modified by Pt, the particle size distribution and average size of nanoparticles of Pt-PdSn-SnO₂ did not obviously change, and the fcc structure of PdSn in all three samples was retained. XPS measurement showed a higher upshift of Pt 4f binding energy occurred for Pt/PdSn-SnO₂ /C (HMTA) compared to those of Pt/PdSn-SnO₂ /C (EDTA) and Pt/PdSn-SnO₂ /C (Nacitrate). Pt/PdSn-SnO₂ /C (HMTA) was also found to have the highest CO and ethanol oxidation activity among the three catalysts.

1. Introduction

The direct ethanol fuel cell (DEFC) affords an attractive alternative to the direct methanol fuel cell (DMFC) because: (1) ethanol is less toxic than methanol and is easier to store and transport due to its relatively higher boiling point; (2) ethanol can be produced in large quantities by chemical or biological processes; and (3) ethanol has higher energy density than methanol [1]. However, implementation of DEFC technology is hindered by the sluggish rate of the ethanol oxidation reaction (EOR) on the anode [2], and thus requires the development of effective anodic catalysts to increase EOR activity.

Recent efforts to develop anodic catalysts for EOR have mainly focused on ternary catalysts based on binary PtSn/C catalysts with various additional elements. Manthiram et al. synthesized carbon-supported PtSnPd electrocatalysts with high EOR activity promoted by oxide phases and the incorporation of Pd into PtSn/C [3]. Spinace et al. showed that substitution of some Sn by Ni resulted in increased current in all potential regions of PtSn/C while hardly shifting the onset potential [4]. They attribute the improved activity of PtSnNi/C to the modification of the electronic structure of Pt and the presence of Ni oxides. Antolini et al. showed that...
PtSnRu/C with different amounts of Ru had higher EOR activity than PtSn/C, attributing this to interactions involving Sn and Ru oxides [5]. Other metals, such as Ir, In and Rh have also been incorporated into PtSn systems and found to enhance catalytic activity [2,6–8]. It should also be noted that the presence of the metal oxidized state in these systems, especially SnO₂, also improved the EOR catalytic activity in the above studies.

In the past two decades, growth of Pt overlayers (or encapsulation by Pt) on other noble metals or alloys have also attracted much attention, providing an effective method to obtain high Pt- mass-based activity at reduced cost [9]. Pt monolayers deposited on carbon-supported Sn, Ni, Ru and Rh nanoparticles have shown greatly enhanced EOR activity [10,11]. Similarly, a SnO₂ core–Pt shell structure has been reported with higher EOR activity than Pt/C and PtSn/C (E-TEK), associating with an increase in the electrochemically active surface area and an electronic effect [12].

It is clear from the above reports that a composite-core catalyst, containing alloy, SnO₂, and a Pt shell structure, affords an effective strategy to enhance catalytic activity. In this case, both the electronic effect from M and Sn in the vicinity of Pt sites, and the bifunctional mechanism from SnO₂, would provide high EOR activity and better poisoning tolerance. Hence, the aim of this work was to prepare Pt/PdSn-SnO₂/C by a sequential reduction process, and to test its performance as an EOR catalyst. The above catalyst was produced using three different stabilizers ethylene diamine tetra-acetic acid (EDTA), sodium citrate (Nacitrate) and hexamethylenetetramine (HMTA) and compared the differences in structure and EOR activity.

2. Experimental
2.1. Synthesis
PdSn-SnO₂/C catalysts, using one of the three different stabilizers, were prepared in three 100 mL flasks containing a solution of 15.6 mg of palladium chloride (PdCl₂), 46.2 mg of stannic chloride pentahydrate (SnCl₄·5H₂O) dissolved in 30 mL of ethylene glycol (EG). Then, 401.8 mg of ethylene diamine tetraacetic acid (EDTA), 288.2 mg of sodium citrate (Nacitrate) and 250.3 mg of hexamethylenetetramine (HMTA) were added to flasks 1–3 respectively, and the solutions were stirred for 0.5 h. 100 mg of pretreated carbon black (Vulcan XC-72R) was added to the mixtures under stirring conditions with ~48 mg of sodium formate and the system adjusted to ~pH 10 by dropwise addition of a 5 wt.% KOH/EG solution. The mixture was maintained at 160 °C for 8 h. The products denoted as PdSn-SnO₂/C (EDTA), PdSn-SnO₂/C (Nacitrate) and PdSn-SnO₂/C (HMTA) were collected by filtration, washed with deionized water four times and dried in air at 60 °C for 5 h.

Pt layers formation used 0.74 mL of H₂PtCl₆·6H₂O (20 mg mL⁻¹) solution dissolved in 20 mL EG with 50 mg of the three above obtained PdSn-SnO₂/C powders
respectively, and the mixtures were stirred for 30 min, followed by addition of 7.5 mg of sodium citrate under constant stirring and adjusted pH to ~10 by the dropwise addition of a 5 wt.% KOH/EG solution. The mixtures were heated at 80 °C for 5 h. The powders were collected by filtration, washed with deionized water to remove chloride from the filtrate, followed by drying in air at 60 °C for 5 h. The Pt modified PdSn-SnO2/C catalysts were denoted as Pt/PdSn-SnO2/C (EDTA), Pt/PdSn-SnO2/C (Nacitrate) and Pt/PdSn-SnO2/C (HMTA), respectively.

2.2 Characterization
X-ray diffraction (XRD) patterns of the catalysts were recorded on a Shimadzu XD-3A diffractometer (Japan) using filtered Cu Kα radiation. Transmission electron microscopy (TEM) measurements were carried out on a JEM-2010 Electron Microscope (Japan) with the acceleration voltage of 200 kV. The average chemical compositions of the catalyst alloy structures were determined using an IRIS advantage inductively coupled plasma atomic emission spectroscopy (ICP-AES) system (Thermo, USA). X-ray photoelectron spectroscopy (XPS) was obtained by a PHI-5702 multifunctional X-ray photoelectron spectrometer (USA).

2.3 Electrochemical measurements
Electrochemical analysis was carried out on a CHI650D electrochemical work station. A conventional three-electrode electrochemical cell was used comprising a platinum counter electrode and an Ag/AgCl (3 mol L⁻¹ KCl) reference electrode. The working electrode was prepared using 5 mg of catalyst dispersed ultrasonically in 1 mL of Nafion/ethanol solution (0.25% Nafion, in volume), where 8 µL of the dispersion was transferred onto a glassy carbon (GC) disc (5 mm in diameter) using a pipette and then air dried. The solution was purged with high-purity N₂ for at least 30 min to ensure gas saturation before each measurement, and stable voltammograms recorded after 10 cycles were taken into account for all the experiments. All electrochemical measurements were carried out at ambient temperature.
3. Results and discussion

In the process of the formation of PdSn-SnO2 nanoparticles, the reaction between the metal precursors, Pd2+ and Sn4+, and the reducing agent is the primary step. At the same time, a competitive reaction to form metal complexes between metal ions and the stabilizers is occurred simultaneously. The rates of these two reactions influence the formation of particle nucleation and growth of nanoparticles as well as the composition of the alloy nanoparticles [13]. In this work, the atoms in coordination with the metal atom of EDTA, sodium citrate and HMTA is O/N, O, and N atom, respectively. Then, the competitive reaction to form metal complexes between Pd and Sn ions and the EDTA, Na citrate and HMTA is occurred differently, which results in the different characterization of PdSn-SnO2 nanoparticles [13–16]. In the second synthesis step, the type and concentration of the stabilizer is same for the all samples. Therefore, even if the procedures of synthesizing nanoparticles are the same, the different PdSn-SnO2 nanoparticles can lead to the different physical characterizations, then which would result in the different electrocatalytic activities.

3.1. Characterization of PdSn-SnO2/C samples

Fig. 1 shows the crystal structure of the prepared PdSn-SnO2 supported on carbon using the three stabilizers, EDTA, Nacitrate and HMTA, characterized by XRD. In Fig. 1a, for comparison, the XRD pattern for a prepared Pd/C sample is also plotted as the vertical dotted line. All the samples show the diffraction peak at ∼24° associated with the (2 0 0) reflection of the hexagonal structure of Vulcan carbon [17], and the characteristic diffraction peaks of Pd (1 1 1) at 2e of ∼40°, Pd (2 0 0) at 2e of ∼46°, Pd (2 2 0) at 2e of ∼67° and Pd (3 1 1) at 2e of ∼81°, which has the face centred cubic (fcc) phase [18].

The reflections associated with the SnO2 phase are located at ∼26° (corresponding to the (1 1 0) plane of tetragonal SnO2), ∼34° (corresponding to the (1 0 1) plane of tetragonal SnO2), ∼52° (corresponding to the (1 1 0) plane of tetragonal SnO2), and ∼42.3° (corresponding to orthorhombic SnO2) [19,20].

While these reflections of the SnO2 phase were not identical, it could be concluded that SnO2 existed in the three samples. The enlarged diffraction patterns in the range of 35–45° are presented in Fig. 1b and show that for PdSn-SnO2/C (HMTA) the (1 1 1) diffraction peaks move to higher positions relative to that of PdSn-SnO2/C (Nacitrate) and PdSn-SnO2/C (EDTA), while 2e of the (1 1 1) plane for PdSn-SnO2/C (EDTA) is slightly larger than that of PdSn-SnO2/C (Nacitrate).
Fig. 2. (a, b and c) TEM images of PdSn-SrO$_2$/C (EDTA), PdSn-SrO$_2$/C (Nacitrato) and PdSn-SrO$_2$/C (HMTA) samples, respectively; and the corresponding nanoparticle size distribution histograms.
Calculated according to the literature [21], the average crystal sizes of the PdSn-SnO$_2$/C (EDTA), PdSn-SnO$_2$/C (Nacitrate) and PdSn-SnO$_2$/C (HMTA) samples were estimated about 5.2, 3.9 and 3.2 nm, respectively. Therefore, the different stabilizers clearly influenced the crystal sizes of PdSn-SnO$_2$/C samples. The lattice parameters [22] were 0.3940, 0.3942, and 0.3934 nm for the PdSn-SnO$_2$/C (EDTA), PdSn-SnO$_2$/C (Nacitrate) and PdSn-SnO$_2$/C (HMTA), respectively. Compared to that of Pd/C (0.3947 nm), the contraction of the Pd crystal lattice in the above samples indicates that Sn entered the crystal lattice of Pd at the atomic level to form PdSn alloy. Due to the similar size atomic radius of Sn (1.72) and Pd (1.79), the contraction was too small to observe the shift of the diffraction peak of the three samples compared to that of Pd/C.

<table>
<thead>
<tr>
<th>Catalyst</th>
<th>a</th>
<th>b</th>
<th>c</th>
</tr>
</thead>
<tbody>
<tr>
<td>Composition/ Pt: Pd: Sn atom ratio</td>
<td>1:2.44:3.60</td>
<td>1:2.42:3.60</td>
<td>1:2.43:3.59</td>
</tr>
<tr>
<td>Mean crystalline size/nm</td>
<td>5.2</td>
<td>4.2</td>
<td>3.2</td>
</tr>
<tr>
<td>Lattice constant/Å</td>
<td>3.003 ± 0.0002</td>
<td>3.897 ± 0.0003</td>
<td>3.875 ± 0.0003</td>
</tr>
<tr>
<td>The average nanoparticle size/nm</td>
<td>5.2</td>
<td>4.2</td>
<td>3.3</td>
</tr>
</tbody>
</table>

Fig. 2 shows TEM images and corresponding particle size distributions of the three samples. For PdSn-SnO$_2$/C (EDTA), as noted in Fig. 2a, the dark spots of PdSn-SnO$_2$ nanoparticles have a spherical morphology and uniform dispersion. The nanoparticle size distribution shown as the inset in Fig. 2a is narrow with an average diameter of 5.2 nm. For PdSn-SnO$_2$/C (Nacitrate), particles had both elliptical and strip morphology, with uniform distribution on the carbon (see Fig. 2b), with a decrease in the size (2.5–5.0 nm) compared to that of the PdSn-SnO$_2$/C (EDTA) sample, with an average diameter of 4.2 nm. For PdSn-SnO$_2$/C (HMTA), Fig. 2c, the Pd-SnO$_2$
nanoparticles were uniformly dispersed on the carbon with irregular spherical shape from 1.5 to 6.5 nm and average diameter of 3.3 nm.

It should be noted that the average nanoparticle size stated was a rough estimation due to the irregular shape of the nanoparticles, especially for the PdSn-SnO$_2$/C (Nacitrate) sample.
3.2. Characterization of Pt/PdSn-SnO\textsubscript{2}/C catalysts

The chemical compositions of the three Pt/PdSn-SnO\textsubscript{2}/C catalysts were evaluated by ICP, listed in Table 1. Based on the concentration of each element, metallic loading was ca. 24 wt.% for the three catalysts.

Fig. 3 shows the XRD patterns of the three Pt/PdSn-SnO\textsubscript{2}/C catalysts. For comparison, the diffraction peak of prepared Pt/C was also plotted as the vertical dotted line. All the samples show the diffraction peak associated with the (2 0 0) plane of Vulcan carbon, and the characteristic diffraction peaks of Pt (1 1 1), (2 0 0), (2 2 0) and (3 1 1), which was present in the fcc phase [23]. For all the catalysts, no diffraction peak related to SnO\textsubscript{2} was found. This was because either some SnO\textsubscript{2} was chemically reduced in the second step of the synthesis procedure, or that the Pt layers on SnO\textsubscript{2} nano-islands had weakened the signals from SnO\textsubscript{2}. However, the presence of SnO\textsubscript{2} in the three Pt/PdSn-SnO\textsubscript{2}/C catalysts was later supported by XPS data.

In addition, compared to the diffraction of home-made Pt/C, the (1 1 1) diffraction peak of the three catalysts slightly shifted towards higher 2\theta values, implying that the crystal lattice of Pt (see Table 1) contracted, which might originate from the strain effect of the core on the Pt layer. Here, the compressive strain shortens the Pt Pt bonds causing d orbital to overlap to a larger degree. Since the d-band of Pt is more than half-filled, the broadening d-band would be expected to result in a downshift in the centre of gravity of the d-band compared to the Fermi level for maintaining constant filling and conserving energy [24,25]. This evidence could also be supported by the following XPS results. On the other hand, the average crystal sizes according to the literature [24] of the three catalysts were estimated and shown in Table 1. A similar average crystal size between the Pt/PdSn-SnO\textsubscript{2}/C catalysts and the PdSn-SnO\textsubscript{2}/C samples was observed, suggesting the Pt layers were thin.

Fig. 4 shows the TEM images of the three Pt/PdSn-SnO\textsubscript{2}/C catalysts and the corresponding particle size distribution histograms. The Pt/PdSn-SnO\textsubscript{2} nanoparticles were well dispersed on the surface of the support with a narrow particle size distribution for the three catalysts. In Table 1, the average nanoparticle size of Pt/PdSn-SnO\textsubscript{2}/C (EDTA), Pt/PdSn-SnO\textsubscript{2}/C (Nacitrate) and Pt/PdSn-SnO\textsubscript{2}/C (HMTA) catalyst does not obviously increase compared to their non-platinized states, suggesting the Pt layer was thin.

Fig. 5 shows the narrow scan of Pt 4f core level of XPS of the three catalysts. The spectra show a doublet containing a low energy band (4f\textsubscript{7/2}) and a high energy band (4f\textsubscript{5/2}), which are listed in Table 2. Here, the Pt 4f binding energy of the prepared Pt/C catalyst is also listed for comparison [26]. For the three Pt/PdSn-SnO\textsubscript{2}/C catalysts, the Pt 4f binding energy gave a positive shift compared to Pt/C. In Pt/PdSn-SnO\textsubscript{2}/C catalysts, the PdSn-SnO\textsubscript{2} core with the smaller lattice parameter introduces strain effects on the Pt shell with the larger lattice parameter [27,28]. The squeezed Pt atoms lead to the increased overlap of the d orbital, and
consequently the band broadens. In order to maintain the same filling degree, the centre of the d-band moves to low energy, resulting in the increase of Pt XPS binding energies, such a phenomenon has been previously suggested [26,29,30].

Furthermore, the stronger strain effect results in the larger shift of Pt XPS binding energy, demonstrated by the consistency between the XPS results and the above XRD results. The upward shifts of the Pt 4f binding energy in Pt/PdSn-SnO2/C catalysts follows in the order Pt/PdSn-SnO2/C (EDTA) ≈ Pt/PdSn-SnO2/C (Nacitrate) < Pt/PdSn-SnO2/C (HMTA), implying that the d-band centre value in Pt/PdSn-SnO2/C (HMTA) catalyst is lower than that of other two catalysts.

The Sn 3d XPS spectra for the three catalysts are shown in Fig. 6. At a glance, the Sn 3d region is more complicated than that of the Pt 4f region. In order to better understand the contribution of different forms of Sn atoms to the different catalysts, deconvolution of XPS Sn 3d peaks was carried out. According to the literature [31–33], the Sn 3d spectrum was deconvoluted using two components, one for Sn(0) and
the second for Sn(IV). For Pt/PdSn-SnO₂/C (EDTA), the Sn 3d⁵/₂ binding energy components are fixed at 485.3 and 486.8 eV (see Table 3), as well as 8.3 eV spin–orbit splitting. The XPS signal at 485.3 eV can be attributed to Sn(0) and that at 486.8 eV to SnO₂ [31,32,34]. The result supports the presence of SnO₂ in Pt/PdSn-SnO₂/C (EDTA) catalyst, which was not discernible previously by XRD analysis. From Fig. 6a, the prevailing occurrence of Sn(IV) in the PdSn-SnO₂ core of the Pt/PdSn-SnO₂/C (EDTA) catalyst, and a atom fraction of Sn(IV) was ca. 68%.

The results of the deconvolution of Pt/PdSn-SnO₂/C (Nacitrate) and Pt/PdSn-SnO₂/C (HMTA) catalysts are also summarized in Table 3. It is evident that, after modification of the PdSn-SnO₂ nanoparticles with the Pt layer, SnO₂ was the predominate form of Sn in the three catalyst. The oxidized forms account for about 55% of the total Sn atom in Pt/PdSn-SnO₂/C (HMTA), which was lower than that of the other two catalysts.

3.3. Electrochemical performance of Pt/PdSn-SnO₂/C catalysts

Fig. 7 shows cyclic voltammetry (CV) results for the oxidation of a monolayer of pre-adsorbed CO (CO stripping voltammograms) obtained for the three catalysts. For clarity, the first scan is shown as a black solid line and the second scan as a red dotted line. A clean surface is recovered in the second cycle coinciding with the voltammogram in the base electrolyte. Common behaviour described in the literature [35] was observed for the three catalysts.

<table>
<thead>
<tr>
<th>Catalyst</th>
<th>Sn 3d⁵/₂/eV</th>
<th>The fraction of Sn(IV)/%</th>
</tr>
</thead>
<tbody>
<tr>
<td>Pt/PdSn-SnO₂/C (EDTA)</td>
<td>485.3, 486.8</td>
<td>68</td>
</tr>
<tr>
<td>Pt/PdSn-SnO₂/C (Nacitrate)</td>
<td>485.3, 486.9</td>
<td>63</td>
</tr>
<tr>
<td>Pt/PdSn-SnO₂/C (HMTA)</td>
<td>485.6, 487.0</td>
<td>55</td>
</tr>
</tbody>
</table>
Fig. 6. XPS of Sn 3d of the three catalysts: (a) Pt/PdSn-SnO₂/C (EDTA), (b) Pt/PdSn-SnO₂/C (Nacitate), and (c) Pt/PdSn-SnO₂/C (HMTA).

Fig. 7. CO stripping voltammograms in 0.5 mol L⁻¹ H₂SO₄ of the Pt/PdSn-SnO₂/C (EDTA), Pt/PdSn-SnO₂/C (Nacitate), and Pt/PdSn-SnO₂/C (HMTA) catalysts, at a potential scan rate of 50 mV s⁻¹.
From Fig. 7, the CO stripping peak potential of the Pt/PdSn-(EDTA) catalyst was almost the same as that of the Pt/PdSn-SnO₂/C (HMTA) catalyst, while the CO stripping peak potential of both of the Pt/PdSn-SnO₂/C (EDTA) and Pt/PdSn-SnO₂/C (HMTA) catalysts were more positive than that of Pt/PdSn-SnO₂/C (Nacitrate), which could be mainly attributed to strain effects. The d-band centre value of Pt in Pt/PdSn-SnO₂/C catalysts was in the order Pt/PdSn-SnO₂/C (HMTA) < Pt/PdSn-SnO₂/C (EDTA) < Pt/PdSn-SnO₂/C (Nacitrate). Here, in correlation with previous reports, Pt surface with a low d-band centre value binds CO more weakly, facilitating CO oxidation [26,36,37]. So, CO oxidation on the Pt/PdSn-SnO₂/C (HMTA) and Pt/PdSn-SnO₂/C (EDTA) electrodes is easy compared to the Pt/PdSn-SnO₂/C (Nacitrate) electrode.

The electrochemically surface area (ECSA) of the catalyst is calculated using the equation [37]:

\[
ECSA = \frac{Q_{CO}}{420W_{Pt}}
\]

where ECSA is the electrochemical active surface area of different catalysts, \( Q_{CO} \) is the charge for CO desorption electro-oxidation in microcoulomb (µC) after subtracting the charge from the 2nd scan region, 420 is the charge required to oxidize a monolayer of CO on the catalyst in µC cm⁻², and \( W \) is the Pd and Pt loading.

Fig. 8. Linear scan voltammetry in 0.5 mol L⁻¹ H₂SO₄ + 0.5 mol L⁻¹ C₂H₅OH + N₂ purged solution with a potential scan rate of 50 mV s⁻¹ recorded at the three catalysts’ electrodes.
The results of ECSA are 27.8, 23.0, and 24.4 m² g⁻¹ for Pt/PdSn-SnO₂/C (HMTA), Pt/PdSn-SnO₂/C (EDTA) and Pt/PdSn-SnO₂/C (Nacitrate) respectively. This result demonstrates that Pt/PdSn-SnO₂/C (HMTA) has more active sites compared with the Pt/PdSn-SnO₂/C (EDTA) and Pt/PdSn-SnO₂/C (Nacitrate) catalysts, which relates to the particle size of Pt/PdSn-SnO₂/C (HMTA) being smaller than that of the other two catalysts.

The EOR activity of the three Pt/PdSn-SnO₂/C catalysts was characterized by anodic sweep cyclic voltammetry (Fig. 8) where the current was normalized for the ECSA of the electrode. The current density of two oxidation peaks for Pt/PdSn-SnO₂/C (HMTA) was 6.3 and 6.7 mA cm⁻² at nearly 0.68 and 1.06 V, respectively, significantly higher than those of Pt@PdSn-SnO₂/C (EDTA) (3.8 and 6.1 mA cm⁻² at ca. 0.67 and 1.06 V, respectively) and Pt/PdSn-SnO₂/C (Na citrate) (3.0 and 4.0 mA cm⁻² at ca. 0.67 and 1.09 V, respectively). The higher current density for Pt/PdSn-SnO₂/C (HMTA) indicates that its performance was superior to Pt/PdSn-SnO₂/C (EDTA) and Pt/PdSn-SnO₂/C (Nacitrate) [38]. In addition, the Pt/PdSn-SnO₂/C (EDTA) catalyst also had higher catalytic activity than the Pt/PdSn-SnO₂/C (Nacitrate) catalyst.

On the basis of the above results, we have shown how different stabilizers affect the distribution and structure of the Pt/PdSn-SnO₂/C catalysts and produce different levels of EOR activity (illustrated as Fig. 9), which we attribute mainly to the strain effect. While the particle size [39,40] and the extent of Sn oxidation [33] may exert an influence on the catalytic activity of Pt/PdSn-SnO₂/C catalysts, this does not explain why Pt/PdSn-SnO₂/C (EDTA) had higher CO and EOR activity than Pt/PdSn-SnO₂/C (Nacitrate).
4. Conclusions
PdSn-SnO₂ nanoparticles supported on Vulcan XC-72 carbon were successfully synthesized using different stabilizers, and modified with Pt layer. Pt/PdSn-SnO₂/C (HMTA) catalyst produced well-separated, smaller, particles (2.2 nm) than those of Pt/PdSn-SnO₂/C (EDTA) and Pt/PdSn-SnO₂/C (Nacitrate), with all three catalysts showing the fcc phase for Pt, PdSn alloy and SnO₂ phase. XPS measurements showed the order of the strain effect to be: Pt/PdSn-SnO₂/C (HMTA) > Pt/PdSn-SnO₂/C (EDTA) > Pt/PdSn-SnO₂/C (Nacitrate). The cyclic voltammetric response for CO and EOR activity revealed that the surface area and the EOR current of the Pt/PdSn-SnO₂/C (HMTA) catalyst was higher than that of the other two catalysts, due to the stronger strain effect on Pt.

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