

CN_x-modified Fe₃O₄ as Pt nanoparticle support for the oxygen reduction reaction

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Abstract

A novel electrocatalyst support material, nitrogen-doped carbon (CN_x)-modified Fe₃O₄ (Fe₃O₄-CN_x), was synthesized through carbonizing a polypyrrole-Fe₃O₄ hybridized precursor. Subsequently, Fe₃O₄-CN_x-supported Pt (Pt/Fe₃O₄-CN_x) nanocomposites were prepared by reducing Pt precursor in ethylene glycol solution and evaluated for the oxygen reduction reaction (ORR). The Pt/Fe₃O₄-CN_x catalysts were characterized by X-ray diffraction, Raman spectra, X-ray photoelectron spectroscopy, scanning electron microscopy, and transmission electron microscopy. The electrocatalytic activity and stability of the as-prepared electrocatalysts toward ORR were studied by cyclic voltammetry and steady-state polarization measurements. The results showed that Pt/Fe₃O₄-CN_x catalysts exhibited superior catalytic performance for ORR to the conventional Pt/C and Pt/C-CN_x catalysts.

Introduction

Among the electrochemical reactions involved in the energy conversion in low-temperature fuel cells, the oxygen reduction reaction (ORR) is the rate-determining process [1]. Platinum is known to possess the best catalytic activity for the ORR among all pure metals when they are deposited on a conductive support material, such as carbon black [2, 3]. Carbon black is commonly used as a support material for Pt-based electrocatalysts, but the corrosion of carbon supports resulting from the electrochemical oxidation in fuel cell operating conditions has been identified to be one of the contributors to the low catalyst performance and durability [4–7].

The ideal support material should have the following characteristics: a high electrical conductivity, easy diffusion of the reactant gas to the electrocatalyst, high enough water absorbent capacity to avoid flooding at the cathode where water is generated, and also showing good corrosion resistance because cathodes are under strongly oxidizing conditions [8]. To enhance the activity of ORR, one of the many strategies is to explore novel materials as catalyst support to host highly active catalysts. Recently, it was found that some oxides, such as SiO₂ [9, 10], Fe₃O₄ [11], and TiO₂ [12], could be used as promising support materials to promote the electrocatalytic stability and even activity of Pt catalysts for ORR. The improved performances depend on the nature of the support. For example, gold particles have a binding energy of about 1.6 eV/defect on an oxygen

vacancy in TiO_2 , which determine the dispersion and shape of the gold particles and further result in an improved catalytic activity [13]. Compared with the amorphous carbon support, Suzuki's group found that the metal-oxide support, such as the WO_3 and TiO_2 , can suppress the Pt lattice expansion and improve the Pt catalytic activity [14]. Among these oxides, it was found that, in conjunction with Fe_3O_4 and Pt, nanoparticles exhibit high catalytic activity for ORR, showing up to 20-fold increase in mass activity compared with the single component Pt nanoparticles and the commercial 3 nm Pt particles. Unfortunately, Fe_3O_4 is an alkaline oxide with low electrical conductivity and easy to dissolve in acidic condition [11]. Fe_3O_4 is therefore difficult to be used directly as electrocatalysts support at low pH conditions. In order to improve the stability and conductivity of Fe_3O_4 , it is necessary to develop an approach to weaken the above-mentioned side effects.

Recently, it has been conclusively shown that forming a good electronically conductive material on the surface of oxides is an effective way to improve the stability and conductivity of oxides [15, 16]. Nitrogen-doped carbon (CN_x) with noble metallic nanoparticles deposited on its surface exhibited an enhanced catalytic activity toward ORR as the CN_x had greater electron mobility and showed n-type or metallic behavior [17–21]. Doping nitrogen into graphite has significant effects on both the morphology and electrocatalytic behavior of Pt nanoparticles [22]. If CN_x can form on the surface of Fe_3O_4 , this new material could be a new desirable catalyst support for electrocatalysts, since Pt will exhibit a good durability on the support surface due to anchoring from nitrogen incorporation [23, 24].

In this study, a novel carbon nitride Fe_3O_4 support ($\text{Fe}_3\text{O}_4\text{-CN}_x$) for Pt electrocatalyst was successfully prepared by carbonizing polypyrrole (PPy, a conducting polymer)-modified Fe_3O_4 ($\text{Fe}_3\text{O}_4\text{-PPy}$) as shown in the Fig. 1. Subsequently, Pt nanoparticles were deposited on $\text{Fe}_3\text{O}_4\text{-CN}_x$ by a modified organic colloid method in an ethylene glycol (EG) solution. Comparing with commercial Pt/C and Pt/C- CN_x , the Pt/ $\text{Fe}_3\text{O}_4\text{-CN}_x$ catalysts exhibited a higher electrocatalytic activity and a better stability for ORR.

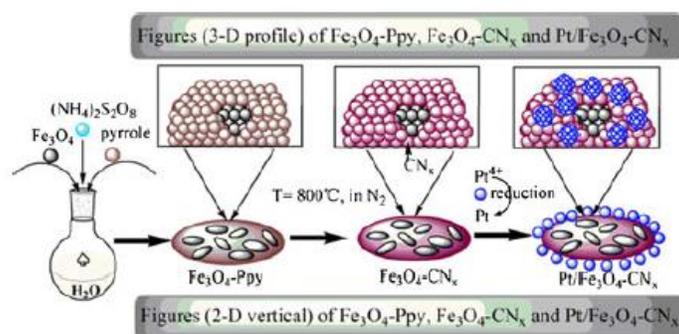


Fig. 1 Scheme of synthesis the new supports and the Pt catalysts

Experimental

Preparation

Fe₃O₄ particles were prepared by the chemical precipitation method. A complete precipitation of Fe₃O₄ was achieved under alkaline conditions. The molar ratio of Fe²⁺/Fe³⁺ was kept at 1:2 under a nonoxidizing environment. To obtain 1 g of Fe₃O₄ precipitate, 0.86 g of FeCl₂·4H₂O and 2.36 g FeCl₃·6H₂O were dissolved in 40 ml of deaerated ultrapure water under a N₂ atmosphere while stirring vigorously (1,000 rpm). The solution was heated to 80 °C, and then 5 ml NH₄OH was added. To ensure the complete growth of the nanoparticle crystals, the reaction was carried out at 80 °C for 30 min under stirring conditions. The resulting suspension was allowed to cool down to room temperature and then washed with ultrapure water to remove any unreacted chemicals. Finally, the resulting wet Fe₃O₄ particles were used for the subsequent adsorption procedure. Polymerization of pyrrole on Fe₃O₄ particles (Fe₃O₄-PPy) was carried out in a solution containing (NH₄)₂S₂O₈ as oxidant. Sixty milliliters of solution containing Fe₃O₄ (0.4 g) and pyrrole monomer (2.0 g) was sonicated at room temperature for 15 min. Eighty milliliters of (NH₄)₂S₂O₈ solution was slowly introduced into the Fe₃O₄/pyrrole monomer suspension. After that, the suspension was kept at 0 °C under stirring conditions for 6 h and then at room temperature for another 12 h. Subsequently, the solid was filtered and washed with ultrapure water at least three times. The obtained black powder (Fe₃O₄-PPy) was dried at 60 °C for 12 h under vacuum. Preparation of Fe₃O₄-CN_x was carried out via the decomposition of Fe₃O₄-PPy, and the procedure was as follows—1 g of Fe₃O₄-PPy were placed in a quartz tube furnace and then heated to 800 °C with a heating rate of 5 °C min⁻¹ under N₂ atmosphere and kept at 800 °C for 2 h. After the furnace cooled down to room temperature, a black powder was obtained. Platinum catalysts (loading ca. 20 wt%) supported on Fe₃O₄-CN_x were prepared by an EG solution method. H₂PtCl₆ (66.4 mg) was dissolved in 30 ml of EG in a flask. The pH of the system was adjusted to 10 by adding 5 wt% of KOH/EG solution. The Fe₃O₄-CN_x (100 mg) black powder was added to the H₂PtCl₆ solution and then heated to 160 °C for 6 h. The resulting catalyst (Pt/Fe₃O₄-CN_x) was filtered, washed with ultrapure water, and dried in a vacuum oven. For comparison purpose, Pt/C-CN_x (20 % Pt, nanoparticles 2.5~6 nm) was synthesized using the same procedure, in which carbon was used to replace Fe₃O₄ as support. Pt/C catalyst were purchased from Johnson Matthey (20 % Pt, particles 3~6 nm).

Materials characterization

X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS) spectra were generated by using a PHI-5702 multifunctional and a PHI-5702 multifunctional X-ray photo-electron spectrometer (American), respectively. Scanning electron microscope (SEM) images were obtained by using a Hitachi S-2400 (Japan). Transmission electron microscopy (TEM) measurements were carried out using a JEM-2010 electron microscope (Japan) with an acceleration voltage of 200 kV. The chemical composition of the samples was determined using the energy dispersive X-ray analysis (EDX) technique coupled to TEM. The elemental

composition of the $\text{Fe}_3\text{O}_4\text{-CN}_x$ and $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ catalysts were evaluated by elemental analysis and EDX. The elemental analysis of carbon (C), nitrogen (N), and sulfur (S) elements were conducted using an organic elemental analyzer (Thermo Flash2000).

Electrochemical characterization

The electrochemical measurements of catalysts were performed on an Autolab electrochemical work station (PGSTAT128N, Eco Chemie, The Netherlands). A conventional three-electrode electrochemical cell was used for the measurements, including a platinum wire as the counter electrode, an Ag/AgCl (KCl 3 M) electrode as the reference electrode and a modified glass carbon electrode (5 mm in diameter, 0.196 cm^2) as the working electrode. All potentials are quoted with respect to the reversible hydrogen electrode (RHE). The thin film electrode was prepared as follows—5 mg of catalyst was dispersed ultrasonically in 1 mL of Nafion/ethanol (0.25 % Nafion). About 8 μL of the dispersion was transferred onto the glassy carbon disc using a pipette and then dried in the air to form catalyst layer on it. Prior to each measurement, the solution was purged with high-purity N_2 (for oxygen-free solutions) or O_2 gas (for oxygen-saturated solutions) for at least 30 min. The Pt-loading on the glass carbon is $40.8\ \mu\text{gPt cm}^{-2}$. All the electrochemical tests were carried out at $25\ ^\circ\text{C}$.

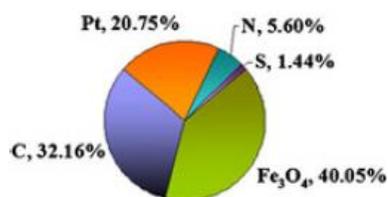


Fig. 2 Contents of $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ catalysts

Results and discussion

Properties of catalysts

The composition contents of C, N, S, Fe_3O_4 , and Pt are shown in Fig. 2. The figure shows that the Pt loading in the $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ catalyst is 20.75 % which is very close to the amount of Pt in the precursors, indicating that the Pt precursor was completely reduced and deposited onto the surface of $\text{Fe}_3\text{O}_4\text{-CN}_x$ catalysts. Since S-containing oxidant ammonium peroxydisulfate was used to prepare $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$, small amount of S was also detected in $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ catalysts.

Vulcan XC-72R, $\text{Fe}_3\text{O}_4\text{-PPy}$, and $\text{Fe}_3\text{O}_4\text{-CN}_x$ were investigated by Raman spectroscopy as shown in Fig. 3. From the figure, it can be observed that the peaks of $\text{Fe}_3\text{O}_4\text{-PPy}$ are similar to the results reported by Liu et al. [25]. Figure 3 also shows that the peaks of $\text{Fe}_3\text{O}_4\text{-PPy}$ disappeared after the carbonization step. Here, the positions and intensities of the peaks can provide information about the structure changes occurring on the surface of the catalysts, namely graphite layer defectiveness, resulting from nitrogen doping [26]. In the curves of

$\text{Fe}_3\text{O}_4\text{-CN}_x$ and the Vulcan XC-72R, two peaks are shown around $1,325$ and $1,591\text{ cm}^{-1}$, indicating the presence of the disorder-band (*D*-band) and tangential-mode peak (*G*-band). The *G*-band indicates the presence of crystalline graphite carbon, and the *D*-band is attributed to disorders in the graphite structure. The intensity ratio of I_D/I_G in $\text{Fe}_3\text{O}_4\text{-CN}_x$ curve is ca. 1.15, which is higher than that of the Vulcan XC-72R (0.99). This finding suggests that there are much more defective sites on the CN_x surfaces formed after nitrogen doping [27]. Raman results also indicate that there are more defect sites available on the structure of $\text{Fe}_3\text{O}_4\text{-CN}_x$. XRD patterns of Fe_3O_4 , $\text{Fe}_3\text{O}_4\text{-CN}_x$, and $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ are shown in Fig. 4.

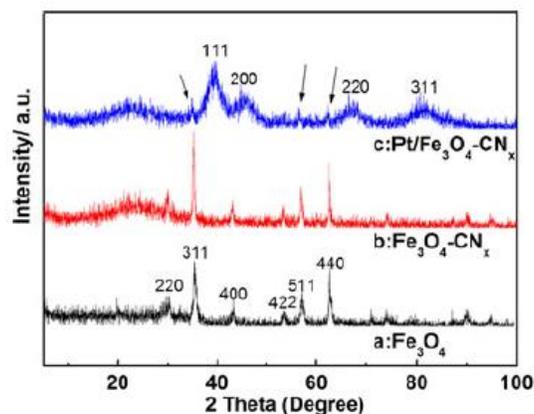
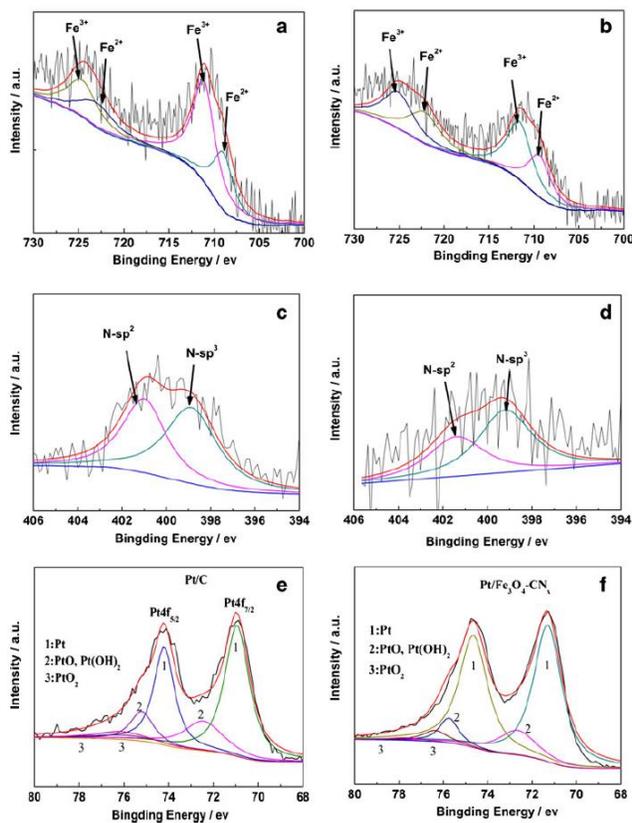


Fig. 4 The XRD patterns of the Fe_3O_4 , $\text{Fe}_3\text{O}_4\text{-CN}_x$, and $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$ catalysts

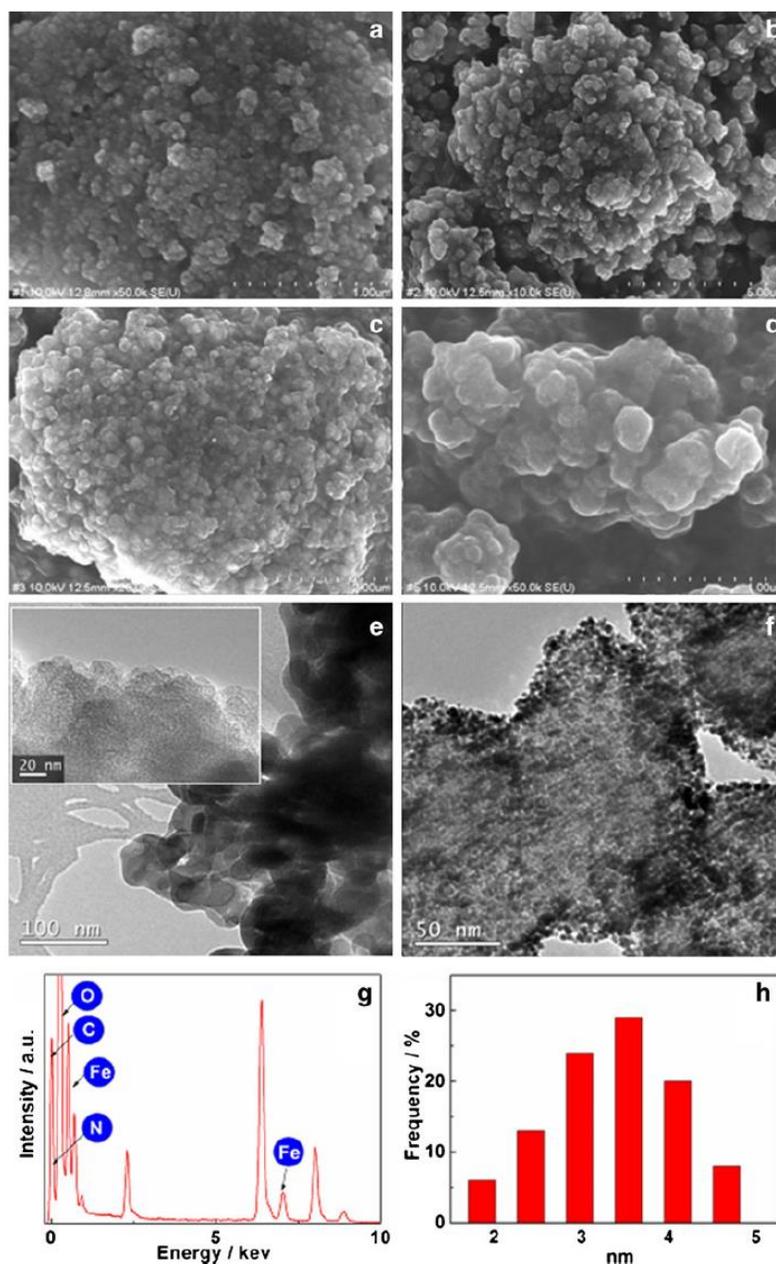
Fig. 5 Fe2p XPS spectra of a $\text{Fe}_3\text{O}_4\text{-CN}_x$ and b $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$; N1s XPS spectra of c $\text{Fe}_3\text{O}_4\text{-CN}_x$ and d $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$; Pt4f XPS spectra of e $\text{Fe}_3\text{O}_4\text{-CN}_x$ and f $\text{Pt/Fe}_3\text{O}_4\text{-CN}_x$



The diffraction peaks of Fe_3O_4 are similar to those found in the literature [28, 29]. A series of characteristic peaks at 30° (2 2 0), 36° (3 1 1), 43° (4 0 0), 53.5° (4 2 2), 57° (5 1 1), and 63° (4 4 0) are well consistent with the data list in JCPDS card no.19-0629. After modification with CN_x , a new broad lump at $2\theta 25^\circ$ (Fig. 4b) is found and assigned to the amorphous CN_x , and all other diffraction peaks are similar to these of Fe_3O_4 particles, which imply that the structure of Fe_3O_4 particles does not change after the carbonization step. Besides, the intensities of the peaks of the Fe_3O_4 phase after the carbonization step are a little bit narrower than that prior the carbonization step. This, in turn, may indicate that the thermal treatment induced the growth of the Fe_3O_4 crystallites. After Pt is deposited on the surface of $\text{Fe}_3\text{O}_4\text{-CN}_x$, the typical diffraction peaks of Fe_3O_4 become relatively small, which may be explained by the relative intensity ration of the Pt, and the $\text{Fe}_3\text{O}_4\text{-CN}_x$ would affect the diffraction peaks. The characteristic peaks of crystalline Pt at $2\theta 39.7^\circ$, 46.3° , 67.6° , 81.4° (JCPDS, no. 04-0802), which are attributed to the (1 1 1), (2 0 0), (2 2 0), and (3 1 1), respectively, are clearly present in the XRD pattern. The XRD results suggest that Pt species were completely reduced to the metallic state by the EG method.

XPS is an efficient method to analyze the surface oxidation states. Figure 5, a and b show the XPS signals of the Fe 2p, as well as their spectral deconvolution curves. Two peaks of Fe 2p_{1/2} at 710.3 eV and Fe 2p_{3/2} at 723.9 eV are observed. The absence of the satellite peak at 719 eV, which is a major characteristic of Fe^{3+} in $\gamma\text{-Fe}_2\text{O}_3$ [30], clearly excludes the formation of $\gamma\text{-Fe}_2\text{O}_3$ in the epitaxial layer. The molar ratio of $\text{Fe}^{2+}/\text{Fe}^{3+}$ obtained from XPS results is almost 1:2, implying that the iron oxide in the catalyst is Fe_3O_4 .

Fig. 6 Scanning electron microscope (SEM) images of Fe_3O_4 (a), Fe_3O_4 -PPy (b), Fe_3O_4 - CN_x (c), and $\text{Pt}/\text{Fe}_3\text{O}_4$ - CN_x (d); transmission electron microscope (TEM) images of Fe_3O_4 - CN_x (e) and $\text{Pt}/\text{Fe}_3\text{O}_4$ - CN_x (f); the *insert* figure of (e) is the edge morphology TEM image of Fe_3O_4 - CN_x ; EDX spectrum of the Fe_3O_4 - CN_x (g); the histograms of particle size distributions of $\text{Pt}/\text{Fe}_3\text{O}_4$ - CN_x (h)



There are four types of nitrogen species in C–N coordination, which are pyridinic (398.6 eV), pyrrolic (400.5 eV), quaternary nitrogen (401.3 eV), and pyridinic $\text{N}^+ - \text{O}^-$ (402–405 eV). Figure 5 shows that there are two peaks in the N 1s spectra for both Fe_3O_4 - CN_x and $\text{Pt}/\text{Fe}_3\text{O}_4$ - CN_x catalysts.

The peak at high-binding-energy, i.e., 401.2 eV, corresponds to nitrogen bonded in a threefold coordination ($\text{N}-sp^2$), which is a feature of N bonding with the graphite layer. The other peak present in the low-binding-energy side corresponds to the pyridine-type N bonding ($\text{N}-sp^3$), which results from the interlinked node morphology inside the CN_x [31]. The area of the XPS peak could be qualitatively estimated the level of element content. Comparing Fig. 5, c and d, the integrated area ratio of $\text{N}-sp^2$ decreased after Pt

nanoparticles were deposited onto the surface, which imply that the amount of N bonded with graphite layer ($N-sp^2$) was decreased after Pt nanoparticles were deposited on the surface.

Figure 5, e and f depict Pt 4f binding energy region of Pt/C and Pt/Fe₃O₄-CN_x catalysts. Two peaks are observed at (71.0, 74.2 eV) for Pt/C and (71.4, 74.8 eV) for Pt/Fe₃O₄-CN_x, which are attributed to 4f_{7/2} and 4f_{5/2} of metallic Pt, respectively. Compared with the XPS of Pt/C, there is a clear shift of the Pt 4f peak in the XPS of the binding energy of Pt/Fe₃O₄-CN_x, shifting to higher energy region, which result from the stronger interaction between Pt and support and forming electron-rich Pt [32]. To identify the surface oxidation states of Pt, the Pt 4f spectrum were deconvoluted into three doublets which correspond to the different oxidation states of Pt. These three pairs of peaks indicate that there are three different oxidation states present in the Pt/C and Pt/ Fe₃O₄-CN_x catalysts. The amount of Pt species is related to the relative intensities of these three peaks. The highest doublet at around 71 and 74 eV is attributed to metallic Pt, which show that the dominant Pt species in the catalysts is metallic Pt.

Figure 6 show SEM images of Fe₃O₄, Fe₃O₄-PPy, Fe₃O₄- CN_x, and Pt/Fe₃O₄-CN_x. It can be observed that Fe₃O₄ particles are aggregated with each other in Fig. 6a. After Fe₃O₄ particles were modified by PPy, a coral-shaped morphology was formed (Fig. 6b). Fe₃O₄-CN_x shown Fig. 6c in also clearly shows a coral-shaped with a quite rough surface after the carbonization step. Figure 6e shows a TEM image of Fe₃O₄-CN_x. Here, it was found that the radius of single coral-like Fe₃O₄-CN_x particle is in the range from 30 to 60 nm. The inserted figure in Fig. 6e represents the edge morphology of Fe₃O₄-CN_x and clearly shows that a rough surface was formed on the Fe₃O₄-CN_x particles, allowing more available sites for immobilizing Pt nanoparticles. The EDX spectrum (as shown in Fig. 6g) indicates that there are C, N, O, and Fe elements present in Fe₃O₄-CN_x. Figure 6, d and f show typical SEM and TEM images of Pt/Fe₃O₄-CN_x catalyst, respectively. It was observed that the Pt nanoparticles were well dispersed on the surface of Fe₃O₄-CN_x support. The diameters of Pt nanoparticles range from 2 to 5 nm which were obtained by measuring 300 particles from Fig. 6f, and the average diameter is 3.6 nm.

Electrochemical characterizations

Figure 7 shows the electrochemical characterizations of the Fe₃O₄-CN_x samples. The cyclic voltammetry (CV) in Fig. 7a shows an oxidation peak (O₁) at ~0.62 V versus RHE in the scan toward positive potential and a reduction peak (R₁) at ~0.50 versus RHE in the negative scan. Both peaks may be related to the overlapped peaks of quinone/hydroquinone peaks and the same Fe(III)/Fe(II) redox system [33]. The stability of the Fe₃O₄-CN_x sample was evaluated by repeated CVs with the appropriate lower and upper potential limits in N₂-purged 0.5 M H₂SO₄ solution at room temperature. The CVs shown in Fig. 7b did not change after 1,500 cycles, indicating that the Fe₃O₄ nanoparticles were

covered by CN_x Figure 8a shows the cyclic voltammograms of Pt/Fe₃O₄-CN_x, Pt/C-CN_x, and Pt/C catalysts in deaerated 0.5 M H₂SO₄ at a scan rate of 50 mVs⁻¹.

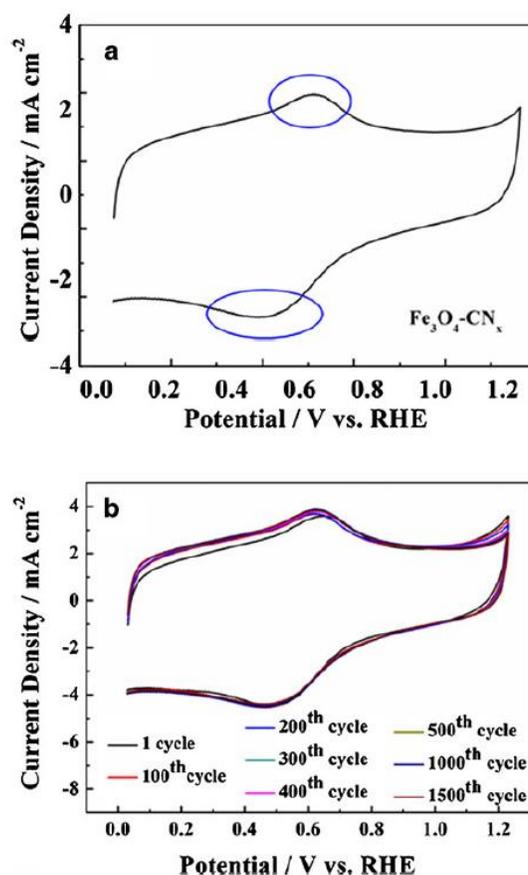


Fig. 7 a CV cycles of the Fe₃O₄-CN_x sample in N₂-purged 0.5 M H₂SO₄ solution at room temperature (sweep rate 50 mV s⁻¹), and b stability of the Fe₃O₄-CN_x sample in N₂-purged 0.5 M H₂SO₄ solution at room temperature (sweep rate 100 mV s⁻¹). In addition, there are some inappropriate expression.

The characteristic peaks in the region (from 0 to 0.3 V versus RHE) are attributed to atomic hydrogen adsorption and desorption on the Pt nanoparticle surfaces. The oxidation and reduction peaks in the curve of Pt/Fe₃O₄-CN_x should be in a similar manner with respect to the curve of Fe₃O₄-CN_x. The change in the reduction peak is due to the quinone/hydroquinone peaks, and the same Fe(III)/Fe(II) reduction peak overlapped with the peak of the Pt oxide reduction. The electrochemical active surface area (*ECSA*) provides important information regarding the number of available active sites on the catalyst. The *ECSA* includes not only the catalyst surface available for charge transfer but also the access of a conductive path to transfer the electrons to and from the electrode surface [34]. The *ECSA* for the catalysts could be estimated from the integrated charge of the hydrogen adsorption/desorption region of the CV. The areas in square meters per gram are calculated from the following formula assuming a correlation value of 0.21 mCcm⁻² and Pt loading [35]:

$$ECSAs(m^2 g_{Pt}^{-1}) = Q_H / (0.21 \times W_{Pt}) \quad (1)$$

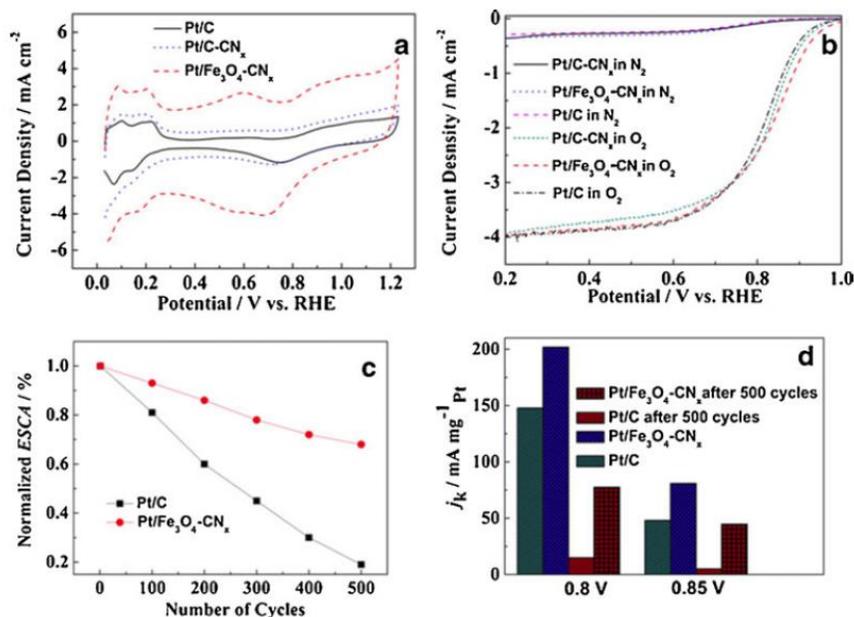
where $ECSAs$ is tQ_H is the amount of charge exchanged during the adsorption of hydrogen atoms on Pt deducted the charge of double layer, W_{Pt} is the Pt loading on the electrode. The $ECSAs$ calculated from the hydrogen adsorption peaks are 14.3, 17.1, and 12.9 $m^2 g_{Pt}$ for Pt/Fe₃O₄-CN_x, Pt/C-CN_x, and Pt/C catalysts, respectively. The $ECSA$ value of Pt/Fe₃O₄-CN_x was found to be 11 % higher than that of conventional Pt/C, indicating that Pt nanoparticles on the Fe₃O₄-CN_x have more electrocatalytic active sites than conventional Pt/C.

Figure 8b shows the linear sweep voltammograms, scanned in the negative-going scan, of Pt/C, Pt/C-CN_x, and Pt/Fe₃O₄-CN_x for ORR. In the Tafel region and the mixed potential region, it was found that the ORR currents of Pt/Fe₃O₄-CN_x and Pt/C-CN_x start to increase at about much higher potentials which are more positive than that of Pt/C, indicating the activity of Pt nanoparticles on the Fe₃O₄-CN_x and C-CN_x supports are higher than that of conventional carbon support. The half-wave potentials of Pt/Fe₃O₄-CN_x, Pt/C, and Pt/C-CN_x are 0.83, 0.81, and 0.82 V versus RHE, respectively, which demonstrate that the catalytic activity of Pt/Fe₃O₄-CN_x catalysts toward the ORR is higher than that of Pt/C and Pt/C-CN_x. The higher activity in electrochemical performance may be related to the following two factors: (1) The Pt/Fe₃O₄-CN_x presented the highest electrochemical surface area, so it is expected to observe an enhanced current in comparison to the Pt/C; (2) the electronic properties are also enhanced due to an extra electron from the doping nitrogen atoms.

The degradation of an electrocatalyst can be evaluated by repeated CVs with the appropriate lower and upper potential limits in an acid solution [36]. The CVs for Pt/C and Pt/Fe₃O₄-CN_x show a significant decrease in platinum $ECSA$ for Pt/C but little for Pt/Fe₃O₄-CN_x as the number of cycles increases, which are shown in Fig. 8c.

Here, the kinetic currents (i_k) were always calculated from the ORR polarization curve by mass-transport correction and normalized to the mass activity of the different catalysts [37], and the mass activity of the different catalysts were shown in the Fig. 8d. Prior to 500 cycles, the Pt/Fe₃O₄-CN_x exhibits a mass activity of 201.8 mA $mg^{-1}Pt$ at 0.80 V (versus RHE) and 80.9 mA $mg^{-1}Pt$ at 0.85 V versus RHE, which is 1.4 and 1.7 times greater than the value obtained for the Pt/C catalyst, i.e., 147.8 mA $mg^{-1}Pt$ at 0.80 V versus RHE and 48.0 mA $mg^{-1}Pt$ at 0.85 V versus RHE.

Fig. 8 a CV cycles of the Pt/Fe₃O₄-CN_x, Pt/C-CN_x, and Pt/C catalysts in N₂-purged 0.5 M H₂SO₄ solution at room temperature (sweep rate, 50 mV s⁻¹). b ORR polarization curves of the Pt/Fe₃O₄-CN_x, Pt/C-CN_x, and Pt/C catalysts in an O₂-saturated 0.5 M H₂SO₄ solution and a N₂-purged 0.5 M H₂SO₄ solution; room temperature; sweep rate, 5 mV s⁻¹; rotation rate, 1,600 rpm. c Loss of electrochemical surface area (*ECSA*) of Pt/C and Pt/Fe₃O₄-CN_x catalysts with number of CV cycles in N₂-purged 0.5 M H₂SO₄ solution at room temperature (0–1.2 V versus RHE, sweep rate 100 mV s⁻¹). d Mass activity at 0.85 and 0.80 V (versus RHE) for Pt/Fe₃O₄-CN_x and Pt/C at the first and the 500 cycles



After 500 cycles, the stability of the catalytic activity was tested by immediately measuring the kinetic currents. It was found that the Pt/Fe₃O₄-CN_x exhibits a high mass activity of 5.2 (0.80 V versus RHE) and 8.7 (0.85 V versus RHE) times greater than those found for the Pt/C catalysts. The enhanced ORR activity of Pt nanoparticles deposited on the Fe₃O₄-CN_x may be attributed to the unique structure and properties of Fe₃O₄-CN_x, which can increase the electron supply from N element and the synergistic effect between Pt and Fe₃O₄-CN_x support when compared with conventional Vulcan XC-72R carbon support. Compared with Pt/C-CN_x, Pt/Fe₃O₄-CN_x presents better stability due to a strong interaction between Fe₃O₄-CN_x support and Pt nanoparticles, which can improve catalyst-support binding [38].

Conclusions

This study shows that Fe₃O₄, an alkaline oxide with low electrical conductivity, is a promising support material for Pt-based ORR electrocatalysts, after it was modified with CN_x. The *ECSA* values of Pt/Fe₃O₄-CN_x was found to be 21 % higher than that of conventional Pt/C. Compared with Pt nanoparticles deposited on carbon, the durability and activity of Pt nanoparticles are greatly enhanced on the Fe₃O₄-CN_x support. The electron from N element and the synergistic effect between Pt and Fe₃O₄-CN_x support result in a high activity towards ORR. Consequently from our study, the Fe₃O₄-CN_x appears to be a promising support material for DMFC electrocatalysts.

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References

1. Xiong LF, More KL, He T (2010) *J Power Sources* 195:2570– 2578
2. Zignani SC, Antolini E, Gonzalez ER (2008) *J Power Sources* 182:83–90
3. Kang YJ, Murray CB (2010) *J Am Chem Soc* 132:7568–7569
4. Huang SY, Ganesan P, Park S, Popov BN (2009) *J Am Chem Soc* 131:13898–13899
5. Gasteiger HA, Kocha SS, Sompalli B, Wagner FT (2005) *Appl Catal, B* 56:9–35
6. Franco AA, (2012) PEMFC degradation modeling and analysis (book chapter). In: Polymer electrolyte membrane and direct methanol fuel cell technology (PEMFCs and DMFCs) vol. 1: fundamentals and performance. C. Hartnig and C. Roth (eds.), Woodhead, Cambridge, UK
7. Franco AA (2008) Gerard M. *J Electrochem Soc* 155:B367–B384
8. Li WZ, Liang CH, Zhou WG, Qiu JS, Zhou ZH, Sun GQ, Xin Q (2003) *J Phys Chem B* 107:6292–6299
9. Miao ZL, Yu HM, Song W, Hao LX, Shao ZG, Shen Q, Hou JB, Yi BL (2010) *Int J Hydrogen Energ* 35:5552–5557
10. Travitsky N, Ripenbein T, Golodnitsky D, Rosenberg Y, Burshtein L, Peled E (2006) *J Power Sources* 161:782–789
11. Wang C, Daimon H, Sun SH (2009) *Nano Letters* 9:1493–1496
12. Kim DS (2010) Abo Zeid EF, Kim YT. *Electrochim Acta* 55:3628– 3633
13. Lopez N, Nørskov JK, Janssens TVW, Carlsson A, Puig-Molina A, Clausen BS, Grunwaldt JD (2004) *J Catal* 225:86–94
14. Suzuki S, Onodera T, Kawaji J, Mizukami T, Yamaga K (2012) *Appl Catal A: General* 427–428:92–97
15. Lee JM, Han SB, Kim JY, Lee YW, Ko AR, Roh B, Hwang I, Park KW (2010) *Carbon* 48:2290–2296
16. Chu YY, Wang ZB, Jiang ZZ, Gu DM, Yin GP (2011) *Adv Mater* 23:3100–3104
17. Tang YF, Allen BL, Kauffman DR, Star A (2009) *J Am Chem Soc* 131:13200–13201
18. Chen Z, Higgins D, Tao HS, Hsu RS, Chen ZW (2009) *J Phys Chem C* 113:21008–21013
19. Shao YY, Sui JH, Yin GP, Gao YZ (2008) *Appl Catal B: Env* 79:89–99
20. Gong KP, Du F, Xia ZH, Durstock ML, Dai LM (2009) *Science* 323:760–764
21. Xiao K, Liu YQ, Hu PA, Yu G, Sun YM, Zhu DB (2005) *J Am Chem Soc* 127:8614–8617
22. Zhou YK, Pasquarelli R, Holme T, Berry J, Ginley D, O’Hayre R (2009) *J Mater Chem* 19:7830–7838
23. Dommele SV, Jong KP, Bitter JH (2006) *Chem Commun* 42:4859– 4861
24. Zamudio AL, Rodríguez-Manzo JA, López-Urías F, Rodríguez-Gattorno G, Lupo F, Rühle M, Smith DJ, Terrones H, Díaz D, Terrones M (2006) *Small* 3:346–350
25. Liu YC, Wang CC (2005) *J Phys Chem B* 109:5779–5782
26. Chen YG, Wang JJ, Liu H, Norouzi Banis M, Li RY, Sun XL, Sham TK, Ye SY, Knights S (2011) *J Phys Chem C* 115:3769–3776
27. Jia NQ, Wang LG, Liu L, Zhou Q, Jiang ZY (2005) *Electrochem Commun* 7:349–354
28. Zhang H, Zhu G (2012) *Appl Surf Sci* 258:4952–4959
29. Padalia D, Johri UC, Zaidi MGH (2012) *Physica B* 407:838–843

30. Zhang DH, Liu ZQ, Han S, Li C, Lei B, Stewart MP, Tour JM, Zhou CW (2004) *Nano Lett* 4:2151–2155
31. Sun CL, Chen LC, Su MC, Hong LS, Chyan O, Hsu CY, Chen KH, Chang TF, Chang L (2005) *Chem Mater* 17:3749–3753
32. Su FB, Tian ZQ, Poh CK, Wang Z, Lim SH, Liu ZL, Lin JY (2010) *Chem Mater* 22:832–839
33. Barrado E, Montequi JI, Medina J, Pardo R, Prieto F (1998) *J Electroanal Chem* 441:227–235
34. Seger B, Kamat PV (2009) *J Phys Chem C* 113:7990–7995
35. Wang R, Zhang Z, Wang H, Lei Z (2009) *Electrochem Commun* 11:1089–1091
36. Chen ZW, Waje M, Li WZ, Yan YS (2007) *Angew Chem* 119:4138–4141
37. Wang W, Wang RF, Ji S, Feng HQ, Wang H, Lei ZQ (2010) *J Power Sources* 195:3498–3503
38. Saha MS, Li RY, Sun XL, Ye SY (2009) *Electrochem Commun* 11:438–441