



# 1D NiHPO<sub>4</sub> nanotubes prepared using dissolution equilibrium as bifunctional electrocatalyst for high-efficiency water splitting

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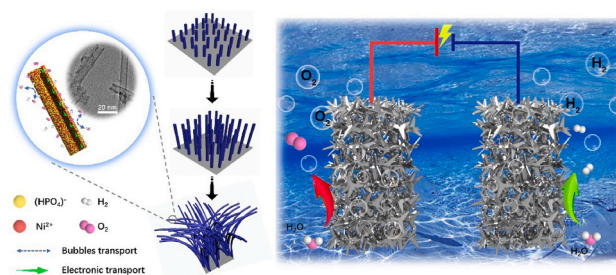
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## HIGHLIGHTS

- Ni foam-supported NiHPO<sub>4</sub> nanotubes catalyst is prepared without any template.
- The mechanism of evolving NiHPO<sub>4</sub> nanotubes based on dissolution equilibrium is proposed.
- NiHPO<sub>4</sub> nanotubes shows good bifunctional performance for overall water splitting
- The present work provides a strategy to explore efficient electrocatalysts.

## GRAPHICAL ABSTRACT



## ARTICLE INFO

### Keywords:

Dissolution equilibrium  
Solvent effect  
Hollow structure  
NiHPO<sub>4</sub> nanotubes  
Water splitting

## ABSTRACT

In this work, one-dimensional NiHPO<sub>4</sub> nanotubes are successfully fabricated on nickel foam by hydrothermal reaction, in which a dissolution equilibrium between phosphates is controlled by tuning the proportion of the mixed solvent and amounts of KOH. As the dissolution equilibrium is broken, the morphology of NiHPO<sub>4</sub> transfers from solid nanowires to hollow nanotubes. The resulting 1D NiHPO<sub>4</sub> nanotubes exhibit good electrocatalytic activity and stability in oxygen evolution reaction (OER) and hydrogen evolution reaction (HER). Notably, a water-splitting voltage of 1.62 V at a current density of 10 mA cm<sup>-2</sup> is obtained in an electrolyzer setup assembled using 1D NiHPO<sub>4</sub> nanotubes as cathode and anode, demonstrating NiHPO<sub>4</sub> nanotubes are promising catalysts for overall water splitting. Moreover, the revealed mechanism of forming tube morphology can be extended to fabricate other metal phosphates with hollow structures.

## 1. Introduction

The global energy crisis and environmental pollution have been

stimulating the development of sustainable energy conversion and storage technologies [1,2]. Hydrogen is now regarded as a promising alternative energy carrier to replace fossil fuels because it can be

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<https://doi.org/10.1016/j.jpowsour.2021.230543>

Received 1 July 2021; Received in revised form 1 September 2021; Accepted 16 September 2021

Available online 24 September 2021

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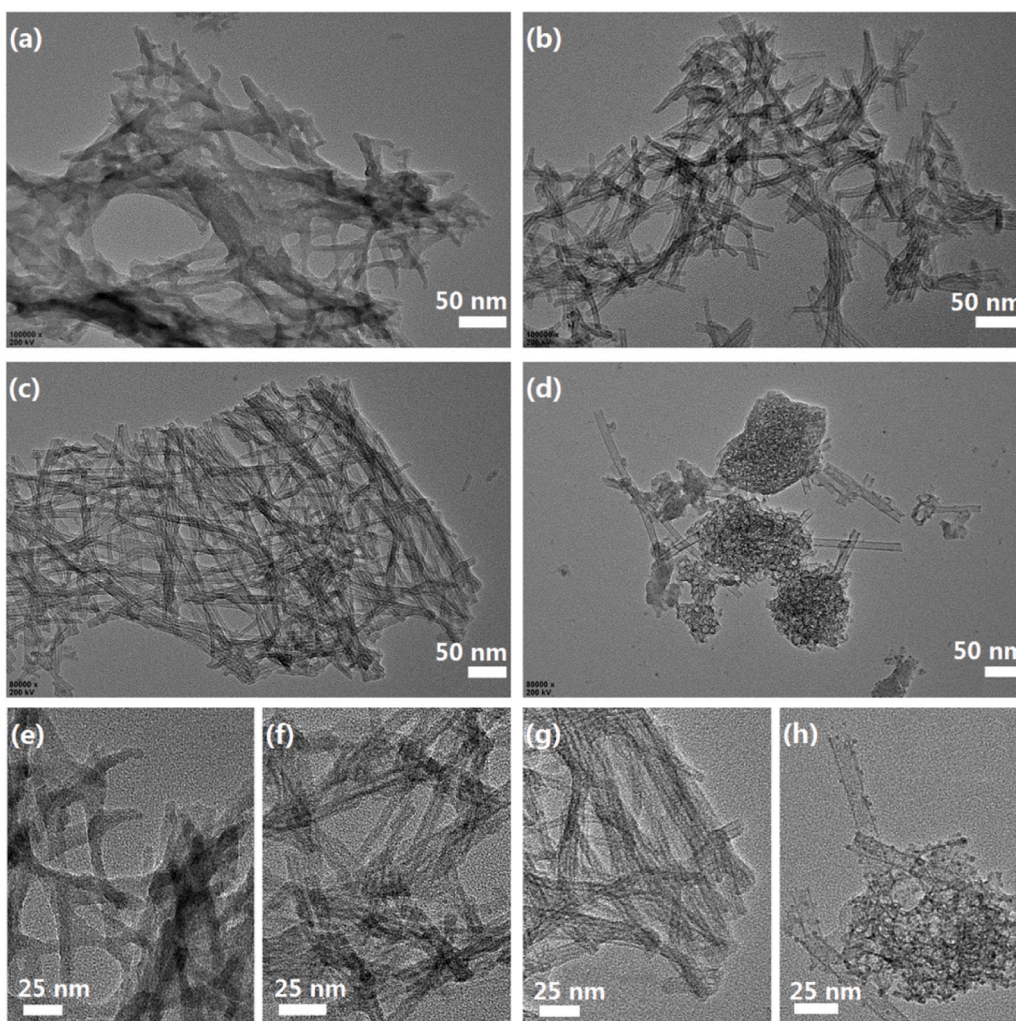


Fig. 1. Effects of the amounts of KOH on the morphology of NiHPO<sub>4</sub>, (a, e) without KOH; (b, f) 0.5 mL KOH; (c, g) 1 mL KOH; (d, h) 5 mL KOH.

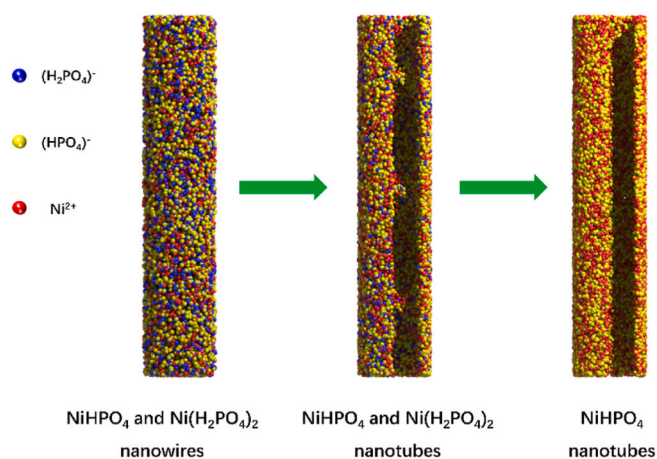
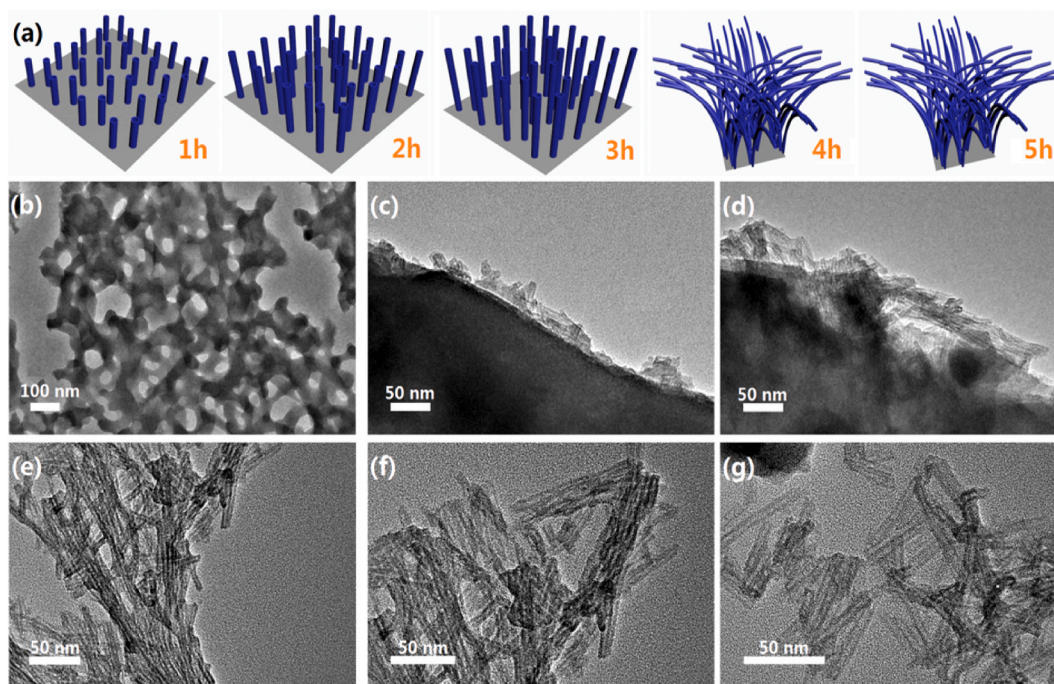


Fig. 2. Schematic diagram of transformation from nanowire to nanotube based on solution equilibrium.

produced by water splitting using excess solar or wind energy. In a water electrolyzer, oxygen evolution reaction (OER) occurs on the anode and hydrogen evolution reaction (HER) on the cathode. Both reactions are occurring in the presence of platinum group metals (PGM) containing electrocatalysts, such as Pt/C for HER, and IrO<sub>2</sub> and RuO<sub>2</sub> for OER,

which reduce the reactions' over-potentials resulting in higher energy conversion efficiency of electrolyzers [3–5]. However, high costs and scarcity of PGM based catalysts significantly impede large scale applications of water electrolysis. Additionally, HER and OER electrocatalysts usually work well in different electrolytes, resulting in low overall efficiency of water electrolysis cells where both cathode and anode are immersed in the same electrolyte [6–8]. Therefore, there is a high demand for bifunctional HER and OER electrocatalysts based on abundant chemical elements to achieve wider adoption of hydrogen generation using water electrolysis.

Many nano-sized transition metal-based materials, such as oxides [9, 10], phosphides [11,12], sulfides [13], nitrides [14,15] and carbides [16,17] have been developed as HER and OER catalysts for water splitting due to their high intrinsic activity and low cost. Recently, transition metal phosphates were added to this group of materials due to their low cost, good electrical conductivity and electrocatalytic performance, which promise significant application potential in electrochemical technologies. Rapid development of nanotechnology and nanostructure engineering provides promising ways to maximize electrocatalytic performance of electrode materials for large-scale applications in renewable and green energy systems. It was reported that charge accumulation and redox reactions can be efficiently improved using one-dimensional (1D) materials, especially 1D nanotubes, due to the space-confined transport phenomena [18,19]. Additionally, the unique hollow structure of 1D nanotubes offers the advantages of high surface area, high porosity and internal void space, providing a large number of



**Fig. 3.** Effects of different reaction times on phosphate structure, (a) NiHPO<sub>4</sub> growth illustration at different heating times; TEM images of the samples after (b) reacted at room temperature for 0.5 h; (c) heated for 1 h; (d) heated for 2 h; (e) heated for 3 h; (f) heated for 4 h; (g) heated for 5 h (all samples shown in Fig. 3c–g were heated at 160 °C).

accessible active sites for surface charge storage and catalytic reactions [20–22]. Moreover, their thin shells could relax structural strains associated with ion insertion [23]. Due to these advantages, 1D nanotubes have become an active research topic in chemical catalysis, biomedical engineering, energy storage and conversion [24]. For instance, 1D hollow structured NiCoP nanowires were synthesized directly on Ni foam using the Kirkendall Effect, and the obtained NiCoP nanowires exhibited low onset potential with superior stability in HER [25]. 1D microrods of NH<sub>4</sub>NiPO<sub>4</sub> were prepared by Ozoemena et al. as an electrode material for pseudocapacitors with high energy rating and power handling [18]. Self-templated techniques to prepare hollow nanostructures are also widely reported, among them processes using hard or soft templates [24, 26–28], advanced selective etching [29,30], outward diffusion [31,32] and heterogenous contraction [33,34]. Although 1D materials show many advantages, their harsh and strictly controlled synthesis conditions still restrict large scale applications, which require the development of easily controllable synthesis routes utilizing abundant resources. Fabricating hollow nanostructures without using any template gained much research attention recently due to simple synthetic procedures and controllable morphology of resulting products. Hollow structures which are made of crystals or noncrystals can be fabricated via the dissolution regrowth process, which, unlike Ostwald ripening, doesn't happen spontaneously, but requires additional chemical to etch the solid particle precursors. During the dissolution regrowth process, the dissolvable precursor redeposits on the surface and results in the formation of hollow structure [23,35].

In this work, motivated by the advantages of transition metal phosphates and 1D nanotubes, we designed and synthesized 1D NiHPO<sub>4</sub> nanotubes using different solubilities of Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> and NiHPO<sub>4</sub>. This was carried out by moderate alteration of the solvent pH to break original hydrolysis balance of H<sub>2</sub>PO<sub>4</sub><sup>-</sup>, and simultaneously synthesize Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> and NiHPO<sub>4</sub>. An investigation into the relationship between the structure of 1D NiHPO<sub>4</sub> nanotubes and their catalytic performance in HER and OER revealed significant improvements in both electrochemical activity and stability. Seldomly reported synthesis of 1D NiHPO<sub>4</sub> nanotubes by controlling the dissolution equilibrium provides a

promising route for the preparation of electrocatalytically active hollow nanomaterials.

## 2. Experiment

**Synthesis of NiHPO<sub>4</sub> nanotubes.** All chemicals used in the study were of analytical grade and utilized directly without further purification. Initially, nickel foam (2 cm × 3 cm) was immersed in a 1.0 M aqueous HCl solution, followed by sonication for 20 min. The acid-treated nickel was washed alternately with ethanol and deionized water, and dried in a vacuum oven at 60 °C. 5 mmol of NaH<sub>2</sub>PO<sub>4</sub> was dissolved in 25 mL of ultrapure water, followed by the addition of 0.5–10 mL 1 M KOH solutions. After the addition of 25 mL 1 mM Ni(NO<sub>3</sub>)<sub>2</sub> ethanol solution and careful placement of the treated Ni foam at their bottom, the containers with NaH<sub>2</sub>PO<sub>4</sub> were transferred into a Teflon-lined autoclave and heated at 160 °C for 1–5 h. The obtained Ni foam was rinsed by ethanol and water and vacuum dried at 40 °C for 12 h.

Reaction equations with adding 0.5, 1.0 mL KOH are as follows:



An appropriate amount of KOH was added to break the original dissolution equilibrium of NaH<sub>2</sub>PO<sub>4</sub> and the hollow structure would form during this process. The details of physical characterization and electrochemical measurements of the synthesized materials are described in the Supporting Information section.

## 3. Results and discussion

### 3.1. Synthesis mechanism of 1D NiHPO<sub>4</sub> nanotubes

**Solvent effect.** Composition and the solubility of the phosphate are

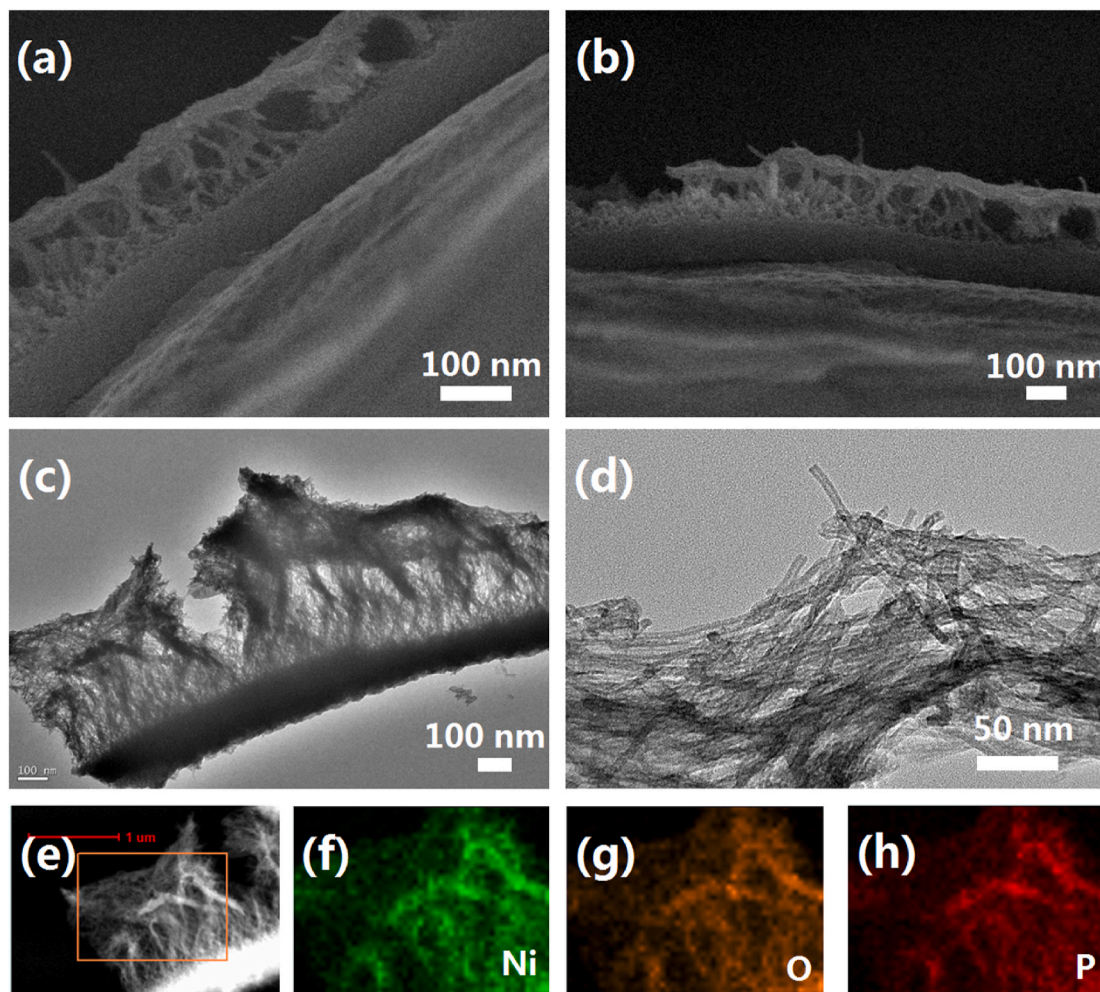


Fig. 4. (a, b) SEM images of NiHPO<sub>4</sub> layer on Ni foam; (c, d) TEM images of NiHPO<sub>4</sub> on Ni foam; (e) STEM image; (f–h) STEM-EDS element maps of Ni, O and P.

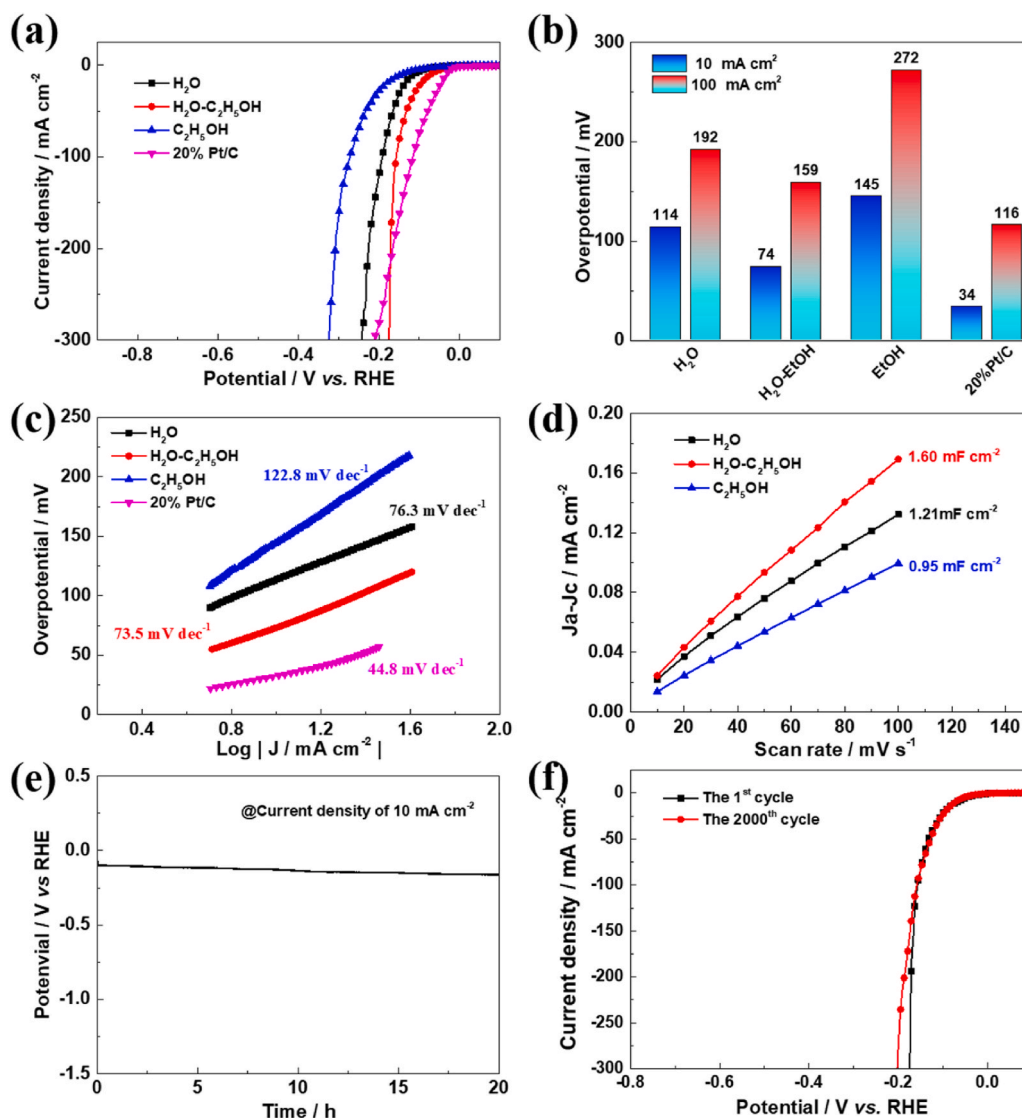
affected by the solvents, then, the effect of the solvent in the NiHPO<sub>4</sub> synthesis process was firstly investigated. As shown in Figs. S1(a and b), the obtained NiHPO<sub>4</sub> consisted of irregular large particles, when only water was used as a solvent during the hydrothermal synthesis. When a mixed solvent containing ethanol and water at a volume ratio of 1:1 was used, a uniform hair-like structure was evenly formed on Ni foam (Figs. S1(c and d)). No NiHPO<sub>4</sub> was detected on Ni foam when only ethanol was used, due to very low precursor solubility in this solvent (Figs. S1(e and f)). These phenomena suggest the composition of the solvents play an important role in the formation of NiHPO<sub>4</sub> nanotubes, in agreement with the previously reported effect of polarity and solubility of solvents on the nucleation and crystal growth of transition-metal-based nanostructures with the possibility of fine-tuning materials' morphology [36].

Similarly, the solvent pH value, determined by KOH concentration, plays a critical role in the type of NiHPO<sub>4</sub> final structure formed. Separate addition of 0, 0.5, 1.0, 5.0, and 10.0 mL of 1 M KOH into ethanol/water mixed solvent with a volume ratio of 1:1, resulted in the preparation of three different materials, such as Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> (without and 0.5 mL KOH), NiHPO<sub>4</sub> (1.0 and 5.0 mL KOH) and Ni<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> (10.0 mL KOH). It can be seen from Figs. S2 and S3 that only solid nanowires originated from the synthesis solutions without KOH addition (Fig. 1(a, e)) and nanotubes were formed in the solutions containing 0.5 mL (Fig. 1(b, f)) and 1.0 mL of KOH (Fig. 1(c,g)). The results indicated the morphology of the products was affected by the solvent pH value.

### 3.2. Dissolution equilibrium process

The above results show that the composition and pH of the solvents play important role in the final structure formed, it could be assumed that the nanotube growth process included several stages, starting with the loss of the hydrolysis balance of H<sub>2</sub>PO<sub>4</sub><sup>-</sup> in ethanol and water mixed solvent at a high pH, resulting in the formation of a supersaturated amorphous phase, rapid nucleation and instantaneous precipitation of mutually converting Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> and NiHPO<sub>4</sub> in the form of stable particles [35]. The conversion of nanowires to nanotubes due to different solubilities of Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> and NiHPO<sub>4</sub> is illustrated in Fig. 2. Dissolution and transformation of Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> leads to the outward diffusion of NiHPO<sub>4</sub> accompanied by voids growth inside the nanowires and its surface re-deposition to produce stable NiHPO<sub>4</sub> nanotubes [35]. No hollow structured nanotubes were observed in corresponding SEM images presented in Fig. S3. when a high quantity of KOH was added to the reaction mixture, indicating that nanotubes can only be obtained at a certain pH range allowing for conversion reactions between Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> and NiHPO<sub>4</sub> to take place. Possible reactions occurring at different amounts of KOH values are proposed in the Supporting Information section.

According to TEM and SEM study of the effect of heating time on the nanotube morphology presented in Figs. 3(b) and S4, only very short and solid nanowires resulted from a 30 min reaction at a room temperature. Heating the reaction mixture at 160 °C for 1 h produced even nanostructured coating, thin enough for Ni foam to be visible in SEM images (Fig. S5). The corresponding TEM image reveals short surface



**Fig. 5.** HER electrocatalytic measurements in 1.0 M KOH: (a) linear sweep voltammogram (LSV); (b) overpotentials comparison for all samples at  $j = 10$  and  $100$  mA cm<sup>-2</sup>; (c) Tafel plots; (d) electrochemical double-layer capacitances; (e) chronopotentiometry of HER at  $10$  mA cm<sup>-2</sup>; (f) HER LSV curves of 1st and 2,000<sup>th</sup> cycles.

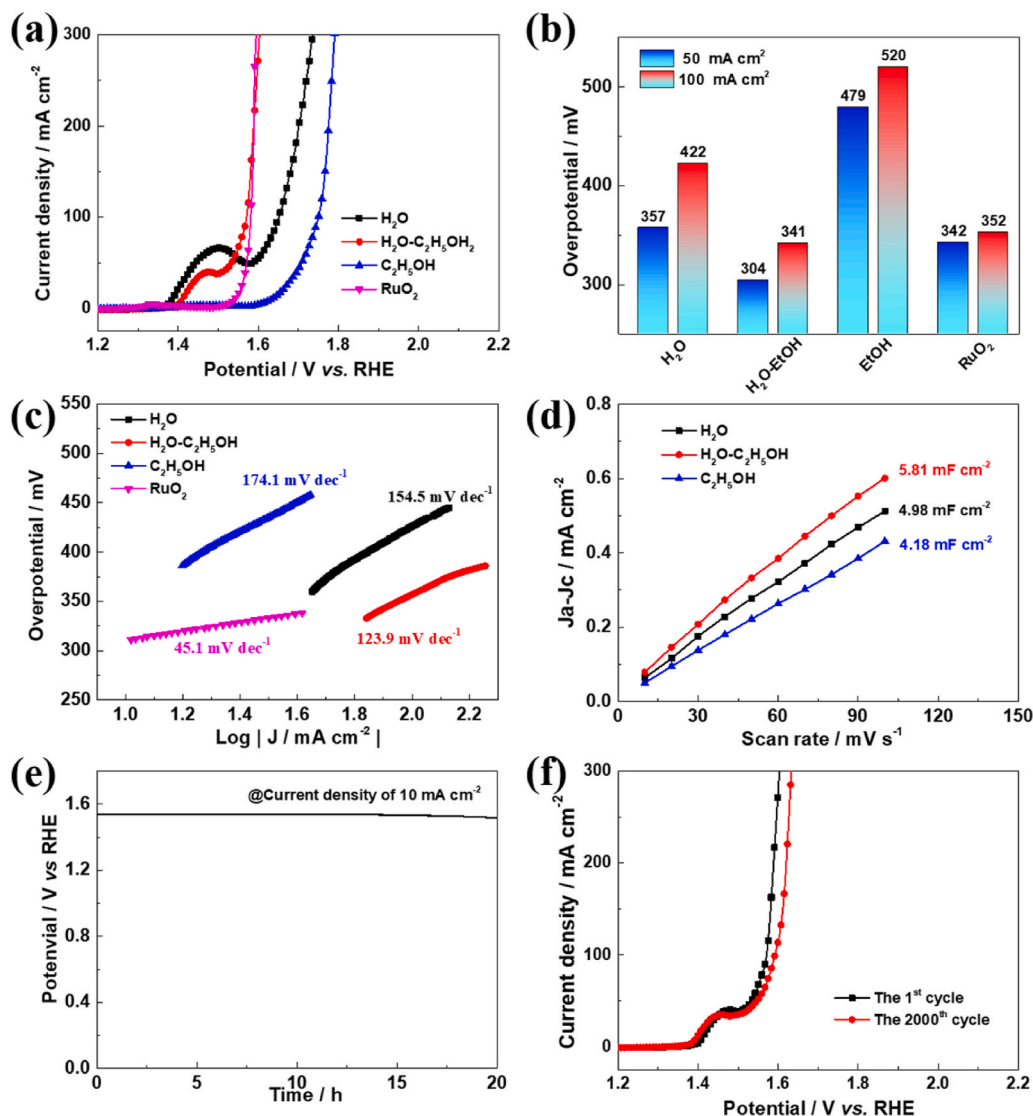
nanotubes whose hollow structure was formed during 1h (Fig. 3(c)). According to Fig. S6, the surface of Ni foam was fully and evenly covered by hair-like nanomaterials when the heating time was further increased to 2 h, and the thickness of the coating increased substantially (Fig. 3 (d)). As the heating time continued to increase, the nanowires gradually grew longer, as shown in Figures S7-8 and 3(e-g). All samples coated during extended periods of time contained nanotubes, as per TEM images (Fig. 3(e-g)), with no structural differences visible for coating times up to 4 h, at which point the process of NiHPO<sub>4</sub> nanotubes formation appeared to be completed.

### 3.3. Characterization of 1D NiHPO<sub>4</sub> nanotubes

The morphology of as-prepared NiHPO<sub>4</sub> coating on the surface of Ni foam was initially investigated by scanning electron microscopy (SEM) which demonstrated a uniform layer with a thickness of ca. 300 nm (Figs. 4(a) and 1(b)). Hair-like structures in the form of thin and dense wires grown vertically on Ni foam are visible in Fig. 4(c). A zoom-in TEM images shown in Figs. 4(d) and S9 reveal that the hair-like layer consisted of straight and long wires with hollow structures of similar diameters, representing a nanotube morphology. No other growths were observed in the TEM image, indicating that uniform NiHPO<sub>4</sub> nanotubes

were formed on Ni foam by the hydrothermal method developed in this study. The scanning transmission electron microscopy (STEM) image and the elemental mapping shown in Fig. 1(f-h) demonstrate even distribution of Ni, O and P elements in the nanotubes, further confirming chemical composition of NiHPO<sub>4</sub>.

Crystallinity of as-prepared NiHPO<sub>4</sub> was analyzed by X-ray diffraction (XRD) as shown in Fig. S10(a). Characteristic peaks of pure Ni at 44.5°, 51.8° and 76.3°, corresponding to (111), (200) and (220) planes respectively (PDF#04-0805), are clearly visible in the XRD pattern of NiHPO<sub>4</sub>. The major characteristic peaks shown in the standard XRD pattern of NiHPO<sub>4</sub> (PDF#39-0706) are well matched with the XRD pattern of the newly prepared material. Besides the diffraction peaks of Ni and NiHPO<sub>4</sub>, no other peaks were observed in Fig. S10(a), confirming that only NiHPO<sub>4</sub> compound was formed on Ni foam. Chemical composition of NiHPO<sub>4</sub> was further evaluated by X-ray photoelectron spectroscopy (XPS) as Fig. S10(b), where a high-resolution Ni 2p spectrum, with two main peaks, corresponding to Ni 2p<sub>3/2</sub> (854.6 eV) and Ni 2p<sub>1/2</sub> (872.4 eV), accompanied by two satellite vibration peaks, is shown. Ni 2p XPS spectrum exhibits a typical feature of Ni<sup>2+</sup> bonded with oxygen [37,38]. The fitted high-resolution P 2p XPS spectrum shows two peaks at 134.0 and 133.1 eV, ascribed to P 2p<sub>3/2</sub> and P 2p<sub>1/2</sub>, due to the oxidized phosphorous species of HPO<sub>4</sub><sup>2-</sup> (Fig. S10(c)) [39].



**Fig. 6.** OER electrocatalytic measurements in 1.0 M KOH: (a) Linear sweep voltammogram (LSV); (b) corresponding overpotentials at  $j = 10, 100 \text{ mA cm}^{-2}$  respectively; (c) Tafel plots; (d) electrochemical double-layer capacitances for the samples; (e) chronopotentiometry measurements of OER at  $10 \text{ mA cm}^{-2}$ ; (f) OER LSV curves for the 1<sup>st</sup> and 2,000<sup>th</sup> cycles of the  $\text{NiHPO}_4$  nanotubes.

**Fig. S10** (d) shows a peak at 529.2 eV corresponding to O 1s, indicating that plenty of oxygen was bonded with P [40] and further confirming the formation of  $\text{NiHPO}_4$ .

### 3.4. Electrochemical performances of 1D $\text{NiHPO}_4$ nanotubes

Electrochemical properties of  $\text{NiHPO}_4$  samples prepared in different solvents were initially evaluated using HER in a 1.0 M KOH electrolyte and benchmarked against the state-of-the-art HER catalyst, Pt/C (20%). As shown in Fig. 5(a), all as-prepared  $\text{NiHPO}_4$  samples demonstrated quite high HER activity. Specifically, the  $\text{NiHPO}_4$  sample prepared in water and ethanol has a low onset potential of ca.  $-23 \text{ mV}$  at a current density of  $10 \text{ mA cm}^{-2}$  (Fig. 5(b)). Although its onset potential was higher than that of Pt/C, the cathode current on  $\text{NiHPO}_4$  increased much faster at higher biases, indicating superior electrochemical activity towards HER [41]. The activity of the catalysts was also normalized by the specific surface area, and the BET surface area was obtained from the nitrogen physisorption provided in Fig. S11. The obtained BET surface area of the samples with adding 0, 0.5, 1, 5, 10 mL were 3.85, 4.71, 5.25, 3.37 and  $3.22 \text{ m}^2 \text{ g}^{-1}$  respectively. The difference of these obtained BET surface areas is quite minimal. This is because nickel foam is the main

part of all the measured samples, and the amount of nickel foam in all the samples is similar. As shown in Fig. S12, the specific activity order of these samples is similar to that of figures presented in Fig. 5(a), in which the current was normalized by the geometric surface area, suggesting the enhanced electrochemical performance is a result of enhanced inherent properties of the catalysts.

Tafel plots were used to further investigate electrochemical properties  $\text{NiHPO}_4$  nanotube-coated materials, as shown in Fig. 5(c) where the sample prepared in  $\text{H}_2\text{O-C}_2\text{H}_5\text{OH}$  has the smallest Tafel value of  $73.5 \text{ mV dec}^{-1}$ , indicating favorable HER reaction kinetics. Cyclic voltammetry (CV) performed in the potential range with no Faradic current (Fig. S13), was instrumental in the evaluation of electrochemical double-layer capacitance, an indication of HER effective surface area, as illustrated in Fig. 5(d), where  $\text{NiHPO}_4$  nanomaterial prepared in  $\text{H}_2\text{O-C}_2\text{H}_5\text{OH}$  solution exhibited the highest values among the three samples.

HER stability of  $\text{NiHPO}_4$  containing bifunctional electrocatalysts, which is critical for their intended water splitting application, was investigated by chronopotentiometry and linear sweep voltammetry (LSV). According to Fig. 5(e), the overpotential of  $\text{NiHPO}_4$  ( $\text{H}_2\text{O-C}_2\text{H}_5\text{OH}$ ) was initially  $-74 \text{ mV}$  and only dropped by  $0.5 \text{ mV}$  after

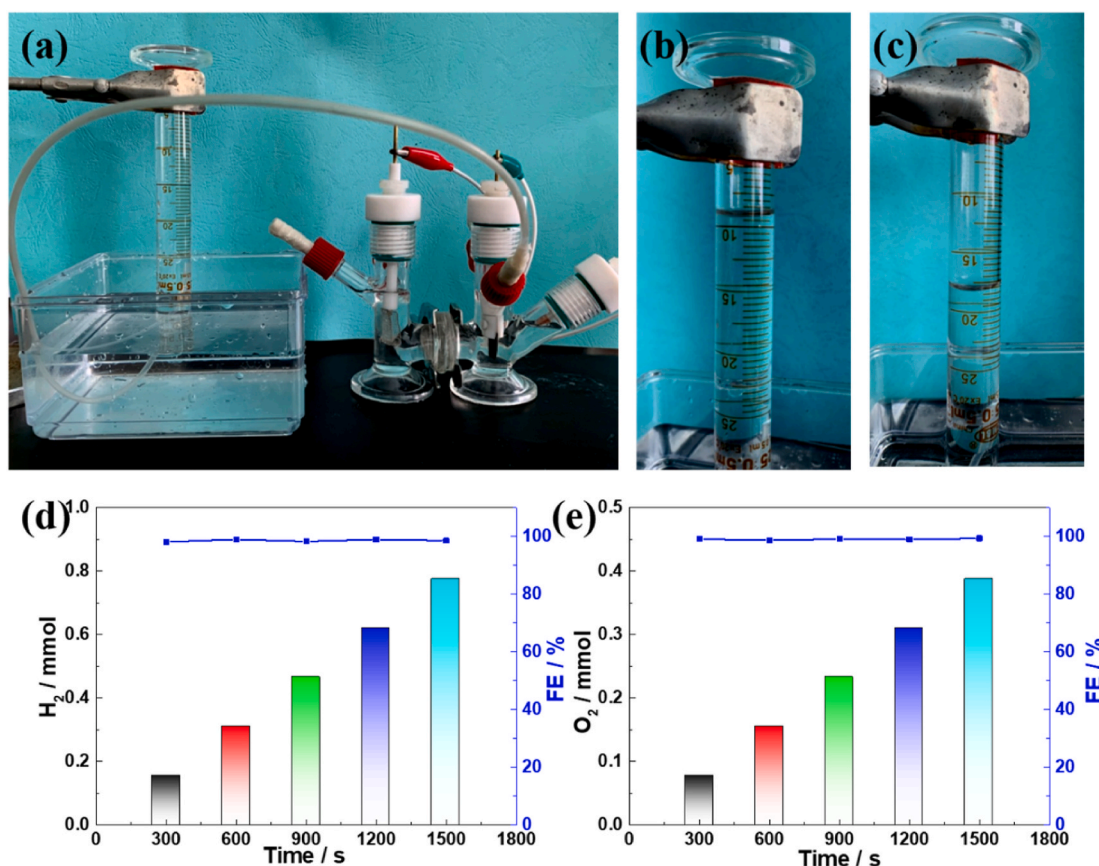


Fig. 7. (a) Three-electrode configuration for gas collection; (b, c) O<sub>2</sub> and H<sub>2</sub> gas volumes collected; (d, e) gas generation rates and corresponding Faradaic efficiencies.

20h of operation at a current density of  $10 \text{ mA cm}^{-2}$ , indicating its good stability under HER conditions. This was further confirmed by LSV plots of 1st and 2000th cycles almost overlapping with each other at current densities lower than  $100 \text{ mA cm}^{-2}$  (Fig. 5(f)). It was also shown that the addition of 1 mL KOH into the NiHPO<sub>4</sub> synthesis mixture resulted in the material with best HER performance among all samples (Figs. S11–15).

The activity of newly prepared catalytic materials toward OER was evaluated in oxygen-saturated KOH electrolyte and compared with that of the state-of-the-art RuO<sub>2</sub> electrocatalyst, which similar performance is illustrated in Fig. 6(a) normalized by the geometric surface area and in Fig. S16 normalized by the specific surface area. While OER overpotentials of NiHPO<sub>4</sub> (H<sub>2</sub>O–C<sub>2</sub>H<sub>5</sub>OH) at 50 and  $100 \text{ mA cm}^{-2}$  were even lower than those of RuO<sub>2</sub>, as shown in Fig. 6(b), NiHPO<sub>4</sub> materials prepared using other solvents were significantly less electrocatalytically active.

Tafel plots were used to study OER kinetics as shown in Fig. 6(c) where, at  $123.9 \text{ mV dec}^{-1}$ , the Tafel slope value of NiHPO<sub>4</sub> (H<sub>2</sub>O–C<sub>2</sub>H<sub>5</sub>OH) is much higher than those of NiHPO<sub>4</sub> prepared in water or ethanol, indicating critical role of nanotubular surface morphology in OER activity of this bifunctional electrocatalyst [31]. This was further confirmed by the evaluation of OER effective surface area using electrochemical double-layer capacitor as shown in Figs. 6(d) and S17. Again, at  $5.81 \text{ mF cm}^{-2}$ , the capacitance of NiHPO<sub>4</sub> (H<sub>2</sub>O–C<sub>2</sub>H<sub>5</sub>OH) was higher than that of other two samples, indicating abundant electrocatalytically active sites available on its surface.

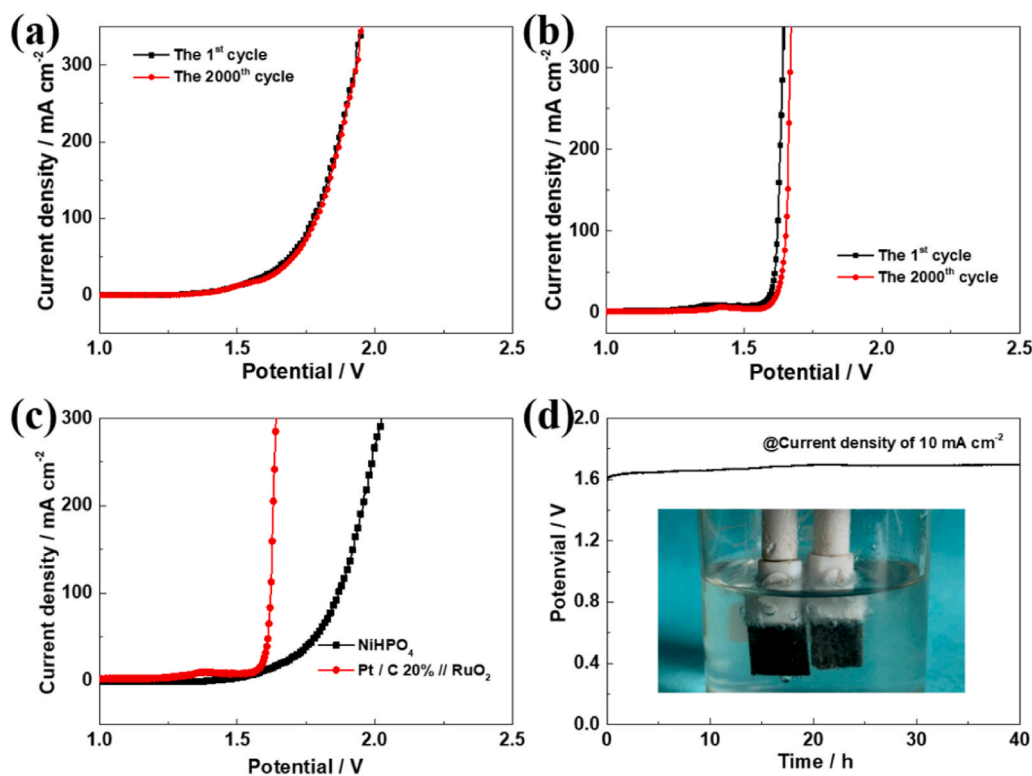
High stability of NiHPO<sub>4</sub> (H<sub>2</sub>O–C<sub>2</sub>H<sub>5</sub>OH) under OER conditions was demonstrated by chronopotentiometry and linear sweep voltammetry as shown in Fig. 6(e–f) where, at a current density of  $10 \text{ mA cm}^{-2}$ , the overpotential decreased by only 30 mV after 20 h operation and shifted positively by 15 mV after 2000 continuous CV cycles. The OER performance investigation was augmented by establishing the optimal

quantity of KOH to be added to the electrocatalyst synthesis mixture, which, according to Figs. S17–19, was 1 mL.

The gas collection electrolysis system shown in Fig. 7(a–c) was used to evaluate the overall water splitting activity of the newly prepared electrocatalyst under more practical conditions. H<sub>2</sub> and O<sub>2</sub> generation were continuously observed at a 2:1 vol ratio (Fig. 7(b and c)) with 0.77 mmol hydrogen collected in about 1500 s, corresponding to hydrogen yield of  $1.85 \text{ mmol h}^{-1}$ , and Faraday efficiency (FE) close to 98.5% (Fig. 7(d)). 0.38 mmol oxygen was collected at the same time at an oxygen production rate of  $0.94 \text{ mmol h}^{-1}$  and FE close to 98.9% (Fig. 7(e)) [42].

Finally, overall water splitting performance of NiHPO<sub>4</sub> nanotube as bifunctional HER and OER electrocatalyst was evaluated under real-world conditions in a two-electrode cell. A current density of  $10 \text{ mA cm}^{-2}$  was achieved at 1.62 V, which was only slightly higher than 1.56 V required to reach the same current density in Pt/C||RuO<sub>2</sub> containing electrolyzer setup (Fig. 8(a–c)). The stable performance of NiHPO<sub>4</sub> electrocatalyst is demonstrated in Fig. 8(d) where 40 h operation of the electrolyzer at  $10 \text{ mA cm}^{-2}$  resulted in a cell voltage increase of only 70 mV, with plenty of gas bubbles evolving on both electrodes as per the insert picture. The performance comparison between NiHPO<sub>4</sub> nanotube electrocatalyst and previously reported Ni based water splitting equivalents, presented in Table S3, clearly demonstrates the advantages of the newly developed material.

The applicability of the obtained NiHPO<sub>4</sub> as anode and cathode in anion exchange membrane water electrolysis (AEMWE), a membrane electrode assembly (MEA) coated with NiHPO<sub>4</sub> were prepared [43,44]. For comparison, the MEA coated with Pt/C and RuO<sub>2</sub> was also fabricated. At the current density of  $100 \text{ mA cm}^{-2}$ , the cell voltage of NiHPO<sub>4</sub> AEMWE cell is 1.94V, with an energy conversion efficiency of 68%, which is close to that of RuO<sub>2</sub> and Pt/C cell. As shown in Table S4, the



**Fig. 8.** (a,b) Polarization curves of the NiHPO<sub>4</sub> nanotubes electrolyzer (a) and Pt/C||RuO<sub>2</sub> (b) for overall water splitting at the 1st and 2,000<sup>th</sup> cycles; (c) polarization curves of NiHPO<sub>4</sub> nanotube electrocatalyst for overall water splitting; (d) chronopotentiometry analysis in two-electrode configuration at constant current density of 10 mA cm<sup>-2</sup>.

electrocatalytic performance of NiHPO<sub>4</sub> AEMWE cell is on par with many other representative transitional-metal-based materials reported previously.

#### 4. Conclusions

In this study, an effective dissolution equilibrium method based on the hydrothermal process was developed to prepare 1D NiHPO<sub>4</sub> nanotubes on Ni mesh. It was found that the formation of 1D NiHPO<sub>4</sub> nanotubes involved in an evolution process from nanoparticle nucleus, solid Ni(H<sub>2</sub>PO<sub>4</sub>)<sub>2</sub> nanofibres to NiHPO<sub>4</sub> nanotubes as a result of breaking the original H<sub>2</sub>PO<sub>4</sub><sup>-</sup> and HPO<sub>4</sub><sup>-</sup> dissolution equilibrium. Using 1D NiHPO<sub>4</sub> nanotube material as both a cathode and an anode in an electrolyzer, a current density of 10 mA cm<sup>-2</sup> could be obtained at 1.62 V, which was attributed to the abundance of active surface sites and enhanced ion transfer properties. Moreover, the NiHPO<sub>4</sub> AEMWE cell also exhibited outstanding performance in 1 M KOH at 25 °C. With high catalytic activity and stability of 1D NiHPO<sub>4</sub> nanotubes grown on Ni foam, this study provides a promising template-free synthesis route towards hollow nanomaterials.

#### CRediT authorship contribution statement

**Zining Wang:** Methodology, Validation, Writing – original draft. **Yutai Wu:** Validation, Writing – original draft. **Mengqi Cui:** Formal analysis. **Shan Ji:** Conceptualization, Resources, Writing – review & editing. **Hui Wang:** Methodology, Validation, Writing – original draft. **Xuyun Wang:** Visualization. **Valdimir Linkov:** Writing – review & editing. **Rongfang Wang:** Conceptualization, Resources, Supervision.

#### Declaration of competing interest

The authors declare that they have no known competing financial

interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Acknowledgments

The authors would like to thank the National Natural Science Foundation of China (21766032), the Key Research and Development Program of Shandong Province of China (2019GGX103029) and the Natural Science Foundation of Shandong Province of China (ZR2020MB024) for financially supporting this work.

#### Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jpowsour.2021.230543>.

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